Chern Number Tunable Quantum Anomalous Hall Effect in Monolayer Transitional Metal Oxides via Manipulating Magnetization Orientation

Zeyu Li,¹ Yulei Han¹,² and Zhenhua Qiao^{1,3,*}

¹CAS Key Laboratory of Strongly-Coupled Quantum Matter Physics, and Department of Physics,

²Department of Physics, Fuzhou University, Fuzhou, Fujian 350108, China

³ICQD, Hefei National Research Center for Physical Sciences at the Microscale,

University of Science and Technology of China,

Hefei, Anhui 230026, China

(Received 9 February 2022; revised 6 May 2022; accepted 27 June 2022; published 14 July 2022)

Although much effort has been made to explore quantum anomalous Hall effect (QAHE) in both theory and experiment, the QAHE systems with tunable Chern numbers are yet limited. Here, we theoretically propose that NiAsO₃ and PdSbO₃, monolayer transitional metal oxides, can realize QAHE with tunable Chern numbers via manipulating their magnetization orientations. When the magnetization lies in the *x-y* plane and all mirror symmetries are broken, the low-Chern-number (i.e., $C = \pm 1$) phase emerges. When the magnetization exhibits nonzero *z*-direction component, the system enters the high-Chernnumber (i.e., $C = \pm 3$) phase, even in the presence of canted magnetization. The global band gap can approach the room-temperature energy scale in monolayer PdSbO₃ (23.4 meV), when the magnetization is aligned to *z* direction. By using Wannier-based tight-binding model, we establish the phase diagram of magnetization induced topological phase transition. Our work provides a high-temperature QAHE system with tunable Chern number for the practical electronic application.

DOI: 10.1103/PhysRevLett.129.036801

Introduction.-As a typical representative of topological phases, the quantum anomalous Hall effect (QAHE) was initially theoretically predicted by Haldane in 1988 [1,2], being characterized by the quantized Hall conductance at zero magnetic field. Because of its dissipationless chiral edge states, the QAHE can be used to build next-generation low-power-consumption electronic devices, and provides a superior platform for investigating novel quantum phenomena such as topological superconductivity and topological magnetoelectric effects [3-6]. Therefore, the realization of QAHE becomes an extraordinarily important topic in the field of condensed matter physics. So far, numerous recipes have been proposed to realize the OAHE [7-19]. The first observation of QAHE is in Cr-doped (Bi, Sb)₂Te₃ thin films at a temperature of 30 mK [20]. Obviously, the low observation temperature is far away from the realistic application. Several works claimed that the magnetic dopants lead to magnetic inhomogeneity, which is detrimental to the formation of QAHE and responsible for the low observation temperature [21–23]. Therefore, the hope of raising the observation temperature becomes to find the intrinsic magnetic topological insulator. Fortunately, in a MnBi₂Te₄ thin film the QAHE was observed at the temperature of 1.4 K, which can further increase to 6.5 K when an external magnetic field is applied to drive the interlayer magnetic coupling from antiferromagnetic to ferromagnetic [24]. This strongly indicates that the intrinsic magnetic insulator may breed high-temperature QAHE for practical application.

In addition to the temperature difficulty, tuning the Chern number to a higher one is another critical issue. High-Chern-number OAHE providing more dissipationless chiral edge channels can significantly improve the performance of quantum anomalous Hall devices [25,26]. So far, several works have shown the theoretical possibility of QAHE with tunable Chern number in various magnetic topological insulator thin films, e.g., by increasing the thin film thickness or doping concentrations [26–30]. The QAHE with tunable Chern number has been realized in thin MnBi₂Te₄ flakes or alternating magnetically doped topological insulator multilayer structures [31,32]. However, these Chern numbers are always fixed for any measured systems. Therefor, it is still challenging and interesting to explore material candidates that can realize QAHE with tunable Chern number for fixed system structures.

In this Letter, we comprehensively investigate the stability, magnetic, electronic, and topological properties of monolayer transitional metal oxides NiAsO₃ and PdSbO₃ crystallized in $P\bar{3}1m$ space group by using first-principles calculations. These materials are predicted to be structurally stable by phonon spectra and molecular dynamics simulations. We show that they are *x*-*y* easy-plane ferromagnetic half-metals with high Berezinskii-Kosterlitz-Thouless critical temperatures (i.e., 216.3 and

University of Science and Technology of China, Hefei, Anhui 230026, China

678.9 K for NiAsO₃ and PdSbO₃, respectively) and wide spin windows (1.26 and 0.97 eV). Six Dirac points appear in the first Brillouin zone due to C_3 and inversion symmetries. After considering spin-orbit coupling, the nontrivial band gaps open at the Dirac points to host QAHE. Remarkably, we find the Chern number C depends on the magnetization orientation, i.e., (i) when the magnetization lies in the *x*-*y* plane and all mirror symmetries are broken, $C = \pm 1$; (ii) when the magnetization has a nonzero *z* component, high-Chern-number $C = \pm 3$ phase arises. We further find that the biaxial strain can efficiently tune the magnetocrystalline anisotropic energy (MAE), making it possible to change the magnetization orientation and realize Chern number tunable QAHE.

Structural properties.—The monolayer NiAsO₃ and PdSbO₃ share the same crystal structure with space group of $P\bar{3}1m$. As displayed in Fig. 1(a), they include five atomic layers, with the transition metal atom being surrounded by six oxygen atoms. The transition metal atoms are in the oxidation state Ni³⁺ and Pd³⁺ with seven valence electrons, respectively. Their lattice constants are listed in Table I. The structural stability is studied by using three different approaches. We first calculate the binding energy expressed as

$$E_b = E(ABO_3) - E(A) - E(B) - \frac{3}{2}E(O_2),$$
 (1)

where $E(ABO_3)$, E(A), E(B), and $E(O_2)$ are the total energies of monolayer NiAsO₃/PdSbO₃, Ni/Pd crystal, As/Sb crystal, and oxygen molecule, respectively. The negative binding energy listed in Table I indicates that its formation is an exothermic reaction. We then perform the phonon spectra calculation with $3 \times 3 \times 1$ supercell,



FIG. 1. (a) Top and side views of monolayer NiAsO₃/PdSbO₃. (b) The first Brillouin zone with high symmetry points, the blue solid lines representing three vertical mirror planes. (c) The bond angle and the schematics of the Ni(Pd)-O-Ni(Pd) super-exchange mechanism in the NiAsO₃ and PdSbO₃ via the (d_{xz}, d_{yz}) -p- (d_{xz}, d_{yz}) orbitals.

where the absence of imaginary phonon frequency strongly suggests the dynamical stability of monolayer NiAsO₃/PdSbO₃. Furthermore, we perform molecular dynamics simulation at 300 K, and the small fluctuations of total energy indicate their thermal stabilities at room temperature (see Fig. S1 in the Supplemental Material [33]). The ternary convex hull diagrams are used to check the thermodynamic stabilities of their bulk counterparts (Ni₂AsO₆ and Pd₂SbO₆). In reality, at monolayer limit, the materials with the same structure have been synthesized, e.g., SrRu₂O₆ [44]. Recently, Sr-terminated ultrathin nanosheets of SrRu₂O₆ have been prepared by using the scalable technique of liquid exfoliation [45], which indicates the possibility to prepare the monolayer NiAsO₃ and PdSbO₃ by the same methods.

Magnetic properties.-To reveal the magnetic ground state of monolayer NiAsO₃ and PdSbO₃, we consider four kinds of magnetic configurations: (i) ferromagnetic, (ii) Néel antiferromagnetic, (iii) stripe antiferromagnetic, and (iv) zigzag antiferromagnetic (see Fig. S3 [33]). The total energies listed in Table I indicate that the ferromagnetic state is preferred in both NiAsO₃ and PdSbO₃. The nearest-neighbor transition metal atoms are linked by oxygen atoms, implying that magnetic ground state is determined by the superexchange mechanism. The bond angle of Ni-O-Ni (Pd-O-Pd) is 91.6° (92.6°) approaching 90°, favoring the ferromagnetic interaction according to Goodenough-Kanamori rules [46,47]. The d orbitals of transition metal atoms surrounded by the distorted octahedra are split into three groups: d_{z^2} , $(d_{xy}, d_{x^2-y^2})$, (d_{xz}, d_{yz}) [48,49]. The projected band structures show that the seven electrons first fill the d_{z^2} and $(d_{xy}, d_{x^2-y^2})$ orbitals, then the remaining one occupies one of the (d_{xz}, d_{yz}) orbitals with a net magnetic moment of 1 μ_B . Because of the strong $pd\sigma$ hybridization between transition metal atoms and oxygen atoms as displayed in Fig. S4 [33], the major exchange interaction is bridged via the orthogonal (p_x, p_y) orbital of oxygen atom, resulting in the ferromagnetic coupling, as shown in Fig. 1(c). The MAE illustrates that the magnetization lies in the x-y plane, i.e., monolayer NiAsO₃ and PdSbO₃ belong to the category of XY magnets (see Fig. S3 [33]). Although the Mermin-Wagner theorem prohibits the long-range magnetic order in two-dimensional isotropic systems, the finite-size effect can stabilize the magnetic order that has been recently verified in easy-plane magnet

TABLE I. Lattice constants, bond angle, binding energy, and total energy E for Néel, stripe, and zigzag antiferromagnetic configurations per unit cell relative to the ferromagnetic ground state in meV.

	a (Å)	c (Å)	θ (°)	E_b (eV)	Néel	Stripe	Zigzag
NiAsO ₃	4.891	3.763	91.6	-1.062	62.7	436.4	173.2
PdSbO ₃	5.276	4.132	92.6	-0.999	93.3	213.4	65.6



FIG. 2. Spin-polarized band structure of monolayer NiAsO₃ (a) and PdSbO₃ (c). (b) Band structure of monolayer NiAsO₃ (b) and PdSbO₃ (d) with spin-orbit coupling when the magnetization along x and z direction, respectively.

1T-VSe₂ and CrCl₃ down to monolayer limit [50–52]. Our estimated critical Berezinskii-Kosterlitz-Thouless transition temperatures for monolayer NiAsO₃ and PdSbO₃ are 216.3 and 678.9 K, respectively [53,54].

Band structures and topological properties.—In Figs. 2(a) and 2(c), one can observe that monolayer NiAsO₃ and PdSbO₃ are half-metals with wide spin windows (1.26 and 0.97 eV, respectively). The spin-down bands demonstrate a large energy gap around the Fermi level, whereas the spin-up bands exhibit metallic electronic structure, implying that they are promising candidates in spintronics such as spin valves and magnetic tunnel junctions [55,56]. Furthermore, along Γ -M high symmetry line, a spin-polarized Dirac point appears at the Fermi level. Because of the C_3 rotation and inversion symmetries, six spin-polarized Dirac points emerge in the first Brillouin zone.

The topological properties of monolayer NiAsO₃ and PdSbO₃ are investigated after turning on the spin-orbit coupling. Because their topological properties are similar we take monolayer PdSbO₃ as an example in the following part (see Fig. S7 for NiAsO₃ [33]). Depending on the orientation of magnetization, various topological phases can be formed. As displayed in Fig. 1(b), there are three mirror planes along Γ -K(K') high symmetry lines related to the C_3 rotation symmetry. When the magnetization is perpendicular to the mirror planes, the spin-polarized Dirac points are still preserved along the Γ -M line that parallels the magnetization direction, leading to a zero Hall conductance (see Fig. S6 [33]). To obtain nonzero anomalous Hall conductivity with in-plane magnetization, it is crucial to break all mirror symmetries [57,58]. When the magnetization is along x direction, as shown in Figs. 3(a)and 3(b), a global gap about 6.6 meV opens in monolayer PdSbO₃ with a Chern number of C = 1. By tuning the orientation of in-plane magnetization, the system can be periodically driven into topological phases with alternative Chern numbers of $C = \pm 1$ in the interval of 60°, e.g., C = -1 phase appears when $\phi = 60^{\circ}$ as displayed in Figs. 3(c) and 3(d). Furthermore, the tiny MAE implies that an external magnetic field can be easily used to align magnetization deviating from the *x*-*y* plane to *z* direction. As displayed in Figs. 3(e)–(h), when the magnetization is aligned to $\pm z$ direction, the monolayer PdSbO₃ will enter a high-Chern-number ($C = \pm 3$) phase with a band gap increase to 23.4 meV.

We further construct a Wannier-based tight-binding model by using (d_{xz}, d_{yz}) orbitals of Pd atom and p orbitals of O atom to reveal the topological phase transition by tuning magnetization orientations. As shown in Fig. S6 [33], when the in-plane magnetization is perpendicular to the mirror plane, i.e., $\phi = 30^{\circ}$ (210°), 90° (270°), 150° (330°), the band gaps are closed with the crossing point located at the Γ -M (M', M") high symmetry line, respectively. Therefore, the Chern number of the system alternatively changes between ± 1 with a period of 60°, as depicted in Fig. 3(j). For the out-of-plane magnetization (e.g., x-z plane), starting from z direction (-z direction), the global band gap gradually decreases with the increase (decrease) of polar angle θ . Combining first-principles calculations, we find the gap will fully closed at $\theta = 66.2^{\circ}$, 78.5°, (101.5°, 113.8°), and then the gap reopens. The critical polar angle determines the phase boundaries of low- and high-Chern-number. Figure 3(1) displays the phase diagram with different Chern numbers, where one can find that the high-Chern-number OAHE can even be realized with a canted magnetization.

The Berry curvatures can also help understand the Chern number variation. In Figs. 3(i) and 3(k), one can clearly see the Berry curvature distribution for in-plane and out-ofplane magnetizations, respectively. For in-plane magnetization, two-thirds anticrossing zones contribute Berry curvatures with the same sign, but opposite in the remaining one-third zone, which is similar to that of LaCl [59]. In contrast, for out-of-plane magnetization, the Berry curvatures give the same contribution at all anticrossing zones. Because each gapped Dirac point contributes $\pm \frac{1}{2}$ Chern number, the in-plane magnetization gives a small net value of low Chern number ($C = \pm 1$) but out-of-plane magnetization a high Chern number ($C = \pm 3$). Hence, possessing multiple Dirac cones in the first Brillouin zone is the prerequisite to realize Chern number tunable QAHE via manipulating magnetization orientation.

Effect of strain.—Because of the flexibility of twodimensional materials, applying strain can effectively tune various material properties [60–64]. We investigate the effect of strain on monolayer NiAsO₃ and PdSbO₃ by applying biaxial strain from -5% to 5%. As displayed in Figs. 4(a) and 4(b), the MAE decreases monotonically from compressive strain to tensile strain, implying that the tensile



FIG. 3. (a),(c): Bulk band structure of monolayer PdSbO₃ along high symmetry line and corresponding anomalous Hall conductivity in the presence of spin-orbit coupling and magnetization lying in *x*-*y* plane for different $\phi = 0^{\circ}$ (a) and $\phi = 60^{\circ}$ (c), respectively. (b),(d): Corresponding energy spectra of semi-infinite ribbon of monolayer PdSbO₃ for different $\phi = 0^{\circ}$ (b) and $\phi = 60^{\circ}$ (d), respectively. Note that there is one gapless edge mode at k = 0 in (b) and (d). (e)–(h) are similar to those in (a)–(d) but with magnetization along *z* and -zdirection, respectively. Note that there are three gapless edge modes near k = 0, π and $-\pi$ in (f) and (h). (i) The distribution of the Berry curvatures for monolayer PdSbO₃ with in-plane magnetization along *x* direction and (k) out-of-plane magnetization along *z* direction, respectively. (j) Phase diagram of Chern number as a function of azimuthal angle ϕ for in-plane magnetization. (l) Phase diagram of Chern number as a function of polar angle θ for the out-of-plane magnetization. The polar radius indicates the value of band gap (in meV).



FIG. 4. (a),(b) Magnetocrystalline anisotropic energy of monolayer NiAsO₃ and PdSbO₃ as a function of applied biaxial strain. (c),(d) Band structure along high symmetry lines, anomalous hall conductivity, and energy spectra of semi-infinite ribbon of PdSbO₃-MoS₂ heterostructure with magnetization along *z* direction.

strain is beneficial to switch the QAHE between low- and high-Chern-number topological phases. It is known that the material growth on certain substrates naturally induces strain. For illustration, we construct a PdSbO₃-MoS₂ heterostructure (1 × 1 PdSbO₃ and $\sqrt{3} \times \sqrt{3}$ MoS₂), in which the PdSbO₃ is sandwiched by MoS₂ (see Fig. S11 [33]). After structural relaxation, the in-plane lattice constant of monolayer PdSbO₃ is enlarged by 3%. The ferromagnetic state and the character of PdSbO₃ are still preserved in this heterostructure. For magnetization along *z* direction, the nontrivial band gap reaches 31.6 meV, exceeding the room-temperature energy scale [see Figs. 4(c) and 4(d)].

Summary.—We systematically investigate the electronic, and topological properties of the stable monolayer transition metal oxides $NiAsO_3$ and $PdSbO_3$ based on the first-principles calculations. Their dynamical and thermal stabilities are confirmed by phonon calculations and molecular dynamics simulations. The magnetization orientation dependent QAHE with low and high Chern number are proposed. The low-Chern-number QAHE is formed for the in-plane magnetization, while the high-Chern-number QAHE arises when the magnetization deviates from *x*-*y* plane to *z* direction. Applying a tensile strain can effectively decrease the MAE in favor of the control for magnetization orientation by an external magnetic field. The tensile strain can be exerted by constructing a heterostructure accompanied by an increasing band gap in the phase with a high Chern number, such as in PdSbO₃-MoS₂ system. Our work provides an ideal platform to realize Chern number tunable QAHE for practical applications.

We are grateful to Prof. Qian Niu for his valuable discussions. This work was financially supported by the National Natural Science Foundation of China (Grants No. 11974327 and 12004369), Fundamental Research Funds for the Central Universities (WK3510000010, WK2030020032), Anhui Initiative in Quantum Information Technologies (Grant No. AHY170000). We also thank the Supercomputing Center of University of Science and Technology of China for providing the high performance computing resources.

*Corresponding author.

qiao@ustc.edu.cn

- K. V. Klitzing, G. Dorda, and M. Pepper, Phys. Rev. Lett. 45, 494 (1980).
- [2] F. D. M. Haldane, Phys. Rev. Lett. 61, 2015 (1988).
- [3] K. He, Y. Wang, and Q.-K. Xue, Annu. Rev. Condens. Matter Phys. 9, 329 (2018).
- [4] X.-L.Qi, T. L. Hughes, and S.-C. Zhang, Phys. Rev. B 82, 184516 (2010).
- [5] X.-L. Qi, T. L. Hughes, and S.-C. Zhang, Phys. Rev. B 78, 195424 (2008).
- [6] D. Xiao, J. Jiang, J.-H. Shin, W. Wang, F. Wang, Y.-F. Zhao, C. Liu, W. Wu, M. H. W. Chan, N. Samarth, and C.-Z. Chang, Phys. Rev. Lett. **120**, 056801 (2018).
- [7] R. Yu, W. Zhang, H. J. Zhang, S. C. Zhang, X. Dai, and Z. Fang, Science **329**, 61 (2010).
- [8] Z. Qiao, S. A. Yang, W.-X. Feng, W.-K. Tse, J. Ding, Y. G. Yao, J. Wang, and Q. Niu, Phys. Rev. B 82, 161414(R) (2010).
- [9] K. F. Garrity and D. Vanderbilt, Phys. Rev. Lett. 110, 116802 (2013).
- [10] Z. Qiao, W. Ren, H. Chen, L. Bellaiche, Z. Zhang, A. H. MacDonald, and Q. Niu, Phys. Rev. Lett. **112**, 116404 (2014).
- [11] C. Fang, M. J. Gilbert, and B. A. Bernevig, Phys. Rev. Lett. 112, 046801 (2014).
- [12] M. M. Otrokov, I. P. Rusinov, M. Blanco-Rey, M. Hoffmann, A. Y. Vyazovskaya, S. V. Eremeev, A. Ernst, P. M. Echenique, A. Arnau, and E. V. Chulkov, Phys. Rev. Lett. **122**, 107202 (2019).
- [13] J. Li, Y. Li, S. Du, Z. Wang, B.-L. Gu, S.-C. Zhang, K. He, W. Duan, and Y. Xu, Sci. Adv. 5, eaaw5685 (2019).
- [14] J. He, X. Li, P. Lyu, and P. Nachtigall, Nanoscale 9, 2246 (2017).

- [15] X.-L. Sheng and B. K. Nikolić, Phys. Rev. B 95, 201402(R) (2017).
- [16] H. P. Wang, W. Luo, and H. J. Xiang, Phys. Rev. B 95, 125430 (2017).
- [17] Z. Liu, Y. Han, Y. Ren, Q. Niu, and Z. Qiao, Phys. Rev. B 104, L121403 (2021).
- [18] G. Chen, A. L. Sharpe, E. J. Fox, Y.-H. Zhang, S. Wang, L. Jiang, B. Lyu, H. Li, K. Watanabe, T. Taniguchi, Z. Shi, T. Senthil, D. Goldhaber-Gordon, Y. Zhang, and F. Wang, Nature (London) 579, 56 (2020).
- [19] M. Serlin, C. Tschirhart, H. Polshyn, Y. Zhang, J. Zhu, K. Watanabe, T. Taniguchi, L. Balents, and A. Young, Science 367, 900 (2020).
- [20] C.-Z. Chang et al., Science 340, 167 (2013).
- [21] I. Lee, C. K. Kima, J. Leea, S. J. L. Billingea, R. Zhong, J. A. Schneelocha, T. Liua, T. Valla, J. M. Tranquada, G. Gua, and J. C. S. Davis, Proc. Natl. Acad. Sci. U.S.A. 112, 1316 (2015).
- [22] E. O. Lachman, A. F. Young, A. Richardella, J. Cuppens, H. R. Naren, Y. Anahory, A. Y. Meltzer, A. Kandala, S. Kempinger, Y. Myasoedov, M. E. Huber, N. Samarth, and E. Zeldov, Sci. Adv. 1, e1500740 (2015).
- [23] S. Qi, Z. Liu, M. Chang, R. Gao, Y. Han, and Z. Qiao, Phys. Rev. B 101, 241407(R) (2020).
- [24] Y. Deng, Y. Yu, M. Z. Shi, J. Wang, X. H. Chen, and Y. Zhang, Science 367, 895 (2020).
- [25] C.-Z. Chang, W. Zhao, D. Y. Kim, P. Wei, J. K. Jain, C. Liu, M. H. W. Chan, and J. S. Moodera, Phys. Rev. Lett. 115, 057206 (2015).
- [26] J. Wang, B. Lian, H. Zhang, Y. Xu, and S. C. Zhang, Phys. Rev. Lett. **111**, 136801 (2013).
- [27] H. Jiang, Z. Qiao, H. Liu, and Q. Niu, Phys. Rev. B 85, 045445 (2012).
- [28] L. Q. Doung, H. Lin, W. F. Tsai, and Y. P. Feng, Phys. Rev. B 92, 115205 (2015).
- [29] J. Zeng, Y. Ren, K. Zhang, and Z. Qiao, Phys. Rev. B 95, 045424 (2017).
- [30] G. Xu, H. Weng, Z. Wang, X. Dai, and Z. Fang, Phys. Rev. Lett. 107, 186806 (2011).
- [31] J. Ge, Y. Liu, J. Li, H. Li, T. Luo, Y. Wu, Y. Xu, and J. Wang, Natl. Sci. Rev. 7, 1280 (2020).
- [32] Y.-F. Zhao, R. Zhang, R. Mei, L.-J. Zhou, H. Yi, Y.-Q. Zhang, J. Yu, R. Xiao, K. Wang, N. Samarth, M. H. W. Chan, C.-X. Liu, and C.-Z. Chang, Nature (London) 588, 419 (2020).
- [33] See Supplemental Material at http://link.aps.org/ supplemental/10.1103/PhysRevLett.129.036801 for the computational methods, magnetic configurations, and MAE; the strain effect on magnetic properties of monolayer NiAsO₃ and PdSbO₃; and the topological properties of monolayer NiAsO₃, which includes Refs. [34–43].
- [34] G. Kresse and J. Furthmüller, Phys. Rev. B 54, 11169 (1996).
- [35] J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. 77, 3865 (1996).
- [36] H. J. Monkhorst and J. D. Pack, Phys. Rev. B 13, 5188 (1976).
- [37] A. Togo and I. Tanaka, Scr. Mater. 108, 1 (2015).
- [38] L. Wang, T. Maxisch, and G. Ceder, Phys. Rev. B **73**, 195107 (2006).

- [39] A. A. Mostofi, J. R. Yates, Y.-S. Lee, I. Souza, D. Vanderbilt, and N. Marzari, Comput. Phys. Commun. 178, 685 (2008).
- [40] X. Wang, J. R. Yates, I. Souza, and D. Vanderbilt, Phys. Rev. B 74, 195118 (2006).
- [41] Q. S. Wu, S. N. Zhang, H.-F. Song, M. Troyer, and A. A. Soluyanov, Comput. Phys. Commun. 224, 405 (2018).
- [42] A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder, and K. A. Persson, APL Mater. 1, 011002 (2013).
- [43] F. Zheng, J. Zhao, Z. Liu, M. Li, M. Zhou, S. Zhang, and P. Zhang, Nanoscale 10, 14298 (2018).
- [44] C. I. Hiley, M. R. Lees, J. M. Fisher, D. Thompsett, S. Agrestini, R. I. Smith, and R. I. Walton, Angew. Chem. Int. Ed. 126, 4423 (2014).
- [45] H. Suvidyakumar, C. Bharat, R. S. Singh, G. Sandeep, D. Tilak, B. Rohit, P. Shivprasad, K. Rdiger, N. Sunil, K. Mukul, and B. Ashna, ACS Appl. Nano Mater. 4, 9313 (2021).
- [46] J.B. Goodenough, Phys. Rev. 100, 564 (1955).
- [47] J. J. Kanamori, J. Phys. Chem. Solids 10, 87 (1959).
- [48] C. Huang, J. Zhou, H. Wu, K. Deng, P. Jena, and E. J. Kan, Phys. Rev. B 95, 045113 (2017).
- [49] S. Mellaerts, R. Meng, V. Afanasiev, J. W. Seo, M. Houssa, and J.-P. Locquet, Phys. Rev. B 103, 155159 (2021).
- [50] S. T. Bramwell and P. C. W. Holdsworth, Phys. Rev. B 49, 8811 (1994).
- [51] H. L. Zhuang and R. G. Hennig, Phys. Rev. B 93, 054429 (2016).
- [52] A. Bedoya-Pinto, J.-R. Ji, A. Pandeya, P. Gargiani, M. Valvidares, P. Sessi, F. Radu, K. Chang, and S. Parkin, Science 374, 616 (2021).

- [53] J. F. Fernández, M. F. Ferreira, and J. Stankiewicz, Phys. Rev. B 34, 292 (1986).
- [54] H. L. Zhuang and R. G. Hennig, Phys. Rev. B 93, 054429 (2016).
- [55] W. E. Pickett and J. S. Moodera, Phys. Today 54, No. 5, 39 (2001).
- [56] I. uti, J. Fabian, and S. D. Sarma, Rev. Mod. Phys. 76, 323 (2004).
- [57] X. Liu, H.-C. Hsu, and C.-X. Liu, Phys. Rev. Lett. 111, 086802 (2013).
- [58] Y. Ren, J. Zeng, X. Deng, F. Yang, H. Pan, and Z. Qiao, Phys. Rev. B 94, 085411 (2016).
- [59] Z. Liu, G. Zhao, B. Liu, Z. F. Wang, J. Yang, and F. Liu, Phys. Rev. Lett. **121**, 246401 (2018).
- [60] Z. Li, Y. Lv, L. Ren, J. Li, L. Kong, Y. Zeng, Q. Tao, R. Wu, H. Ma, B. Zhao, D. Wang, W. Dang, K. Chen, L. Liao, X. Duan, X. Duan, and Y. Liu, Nat. Commun. 11, 1151 (2020).
- [61] Z. Liu, M. Amani, S. Najmaei, Q. Xu, X. Zou, W. Zhou, T. Yu, C. Qiu, A. G. Birdwell, F. J. Crowne, R. Vajtai, B. I. Yakobson, Z. Xia, M. Dubey, P. M. Ajayan, and J. Lou, Nat. Commun. 5, 5246 (2014).
- [62] C. Zhang, M.-Y. Li, J. Tersoff, Y. Han, Y. Su, L.-J. Li, D. A. Muller, and C.-K. Shih, Nat. Nanotechnol. 13, 152 (2018).
- [63] G. H. Ahn, M. Amani, H. Rasool, D. H. Lien, J. P. Mastandrea, J. W. Ager, M. Dubey, D. C. Chrzan, A. M. Minor, and A. Javey, Nat. Commun. 8, 608 (2017).
- [64] Z. Dai, L. Liu, and Z. Zhang, Adv. Mater. 31, 1805417 (2019).