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IOA Posting Date: November 13, 2020

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DOI: 10.31635/ccschem.020.202000555

# Synthesis of iron carbide nanoparticles: Identification of the active phase and mechanism of Fe-based Fischer-Tropsch Synthesis

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# **Abstract**

Despite the extensive study of Fe-based Fischer-Tropsch synthesis (FTS) over the past 90 years, its active phases as well as the reaction mechanisms are still unclear due to the co-existence of metals, oxides and carbide phases presented under realistic FTS reaction conditions and the complex reaction network involving CO activation, C-C coupling and methane formation. To address this issue, we successfully synthesized a range of pure phase iron and iron carbide nanoparticles (Fe, Fe<sub>5</sub>C<sub>2</sub>, Fe<sub>3</sub>C,

 $Fe_7C_3$ ) for the first time. By using them as the ideal model catalysts on high-pressure transient experiments, we identified unambiguously that all the iron carbides are catalytically active in FTS reaction while  $Fe_5C_2$  is the most active yet stable carbide phase, which is consistent with density functional theory (DFT) calculation results. The reaction mechanism and kinetics of Fe-based FTS was further explored on the basis of those model catalysts by means of transient high-pressure stepwise temperature-program surface reaction (STPSR) experiment and DFT calculations. Our work provides new insights into the active phase of iron carbides and corresponding FTS reaction mechanism, which is essential for better iron-based catalysts design for FTS reaction.

# **Keywords**

Fischer-Tropsch synthesis, iron carbides, active phase, mechanism

#### Introduction

Fischer-Tropsch synthesis (FTS) receives increasing attention because synthesis gas (syngas) can be obtained from coal, biomass and shale gas, which becomes particularly importance as alternative fuel and chemicals production. Despite the extensive exploration of Fe-based catalysts over the past 90 years, its active phases and reaction mechanisms are still in controversial. The typical iron catalysts, usually produced from thermal reduction and successive activation of iron oxide precursors, contain different phases including metals, oxides and carbides produced during the pretreatment of the catalysts by carbon-containing gases such as CO.4-10 The phase evolution of the iron catalysts during FTS reaction is even more complicated, and in most cases, a mixture of different phases was resulted during the FTS reaction. 11-15 To address this issue, various types of metal or metal carbides were prepared. Metallic iron, 23-24 and various phases of iron carbides had all been claimed to be active. A fo. 25-38 The complication for FTS comes also from the complexity of the reaction network itself, including CO activation, C-C bond formation and methane formation. A however, the reaction

performance dependence on catalyst phases of iron and carbides as well as the catalytic mechanism behind was still elusive, which hinders better Fe-based catalyst design for FTS.

To address these challenging issues, we have successfully, for the first time, synthesized a variety of pure phase iron and iron carbide catalysts with similar particle size, including Fe, Fe<sub>7</sub>C<sub>3</sub>, Fe<sub>3</sub>C, Fe<sub>5</sub>C<sub>2</sub>, etc., which allow us to identify their initial and intrinsic activities as well as the structural evolutions of iron-based catalysts during FTS. At the same time, we developed a new transient experiment, i.e., transient high-pressure stepwise temperature-program surface reaction (STPSR), which enables us to explore directly the complicated yet challenging problems in FTS such as the fundamental knowledge about syngas activation, hydrocarbon and methane formation on these pure phase catalysts, which were not available before. The comprehensive density functional theory (DFT) calculations revealed a deep insight on the intrinsic activity of iron metal and iron carbides on CO activation, C-C bond and methane formation, rationalizing the kinetic and thermodynamic origin on the structural evolution of different iron-based catalysts during the FTS reaction. The deeper understanding the active phase of iron-based catalysts and corresponding FTS reaction mechanism is beneficial for rational design of more effective Fe-based catalysts on FTS by the synthesis of more  $Fe_5C_2$  catalyst.

#### **Experimental Methods**

# Synthesis of Fe<sub>7</sub>C<sub>3</sub> and Fe<sub>2</sub>C NPs

In a four-neck flask, 20 mL of N,N-dimethyloctadecylamine (for Fe<sub>7</sub>C<sub>3</sub>) or dodecylamine (for Fe<sub>2</sub>C) was stirred sufficiently and degassed under 120 °C for 2 h. Then, the system was refilled with NH<sub>3</sub> and heated to 180 °C. After that, Fe(CO)<sub>5</sub> (0.7 mL, 5.0 mmol) was injected under NH<sub>3</sub> atmosphere and kept at this temperature for 30 min. A colour change from orange to black was observed during the process, implying the decomposition of Fe(CO)<sub>5</sub> and the nucleation of Fe nanocrystals. Subsequently, the mixture was further heated to 350 °C (for Fe<sub>7</sub>C<sub>3</sub>) or 260 °C (for Fe<sub>2</sub>C) at 10 °C /min and kept for 2

h before it was cooled down to room temperature. The product was washed with ethanol and hexane, and collected for further characterization. The as-synthesized NPs were kept in Ar-filled glove box to avoid exposure to air before further characterization. In the absence of NH<sub>3</sub>, the nanoparticles (NPs) would be oxidized to iron oxide (Fig S1 and Fig. S2).

#### Synthesis of Fe<sub>5</sub>C<sub>2</sub> and Fe NPs

In a four-neck flask, a mixture of octadecylamine (14.5 g) and CTAB (0.113 g) was stirred sufficiently. Then, the system was refilled with  $N_2$  and heated to 180 °C. Following that,  $Fe(CO)_5$  (0.5 mL, 3.6 mmol) was injected under a  $N_2$  blanket. The mixture was kept at 180 °C for 10 min. A color change from orange to black was observed during the process, implying the decomposition of  $Fe(CO)_5$  and the nucleation of Fe nanocrystals. Subsequently, the mixture was further heated to 350 °C (for  $Fe_5C_2$ ) or 300 °C (for Fe) at 10 °C /min and kept for 10 min before it was cooled down to room temperature. The product was washed with ethanol and hexane, and was kept in Ar-filled glove box to avoid exposure to air before further characterization.

#### Preparation of supported catalyst

The NPs obtained from high temperature liquid phase synthesis had been washed with n-hexane and ethanol for several times and dispersed in n-hexane under  $N_2$  protection. After, the dispersion of iron carbide NPs was added into a certain amount of silica ( $N_2$  adsorption-desorption isotherm was shown in Fig. S3) under stirring. After evaporating the solvent at room temperature, the supported catalyst was prepared. The amount of iron determined by ICP was around 8%.

#### **Catalysis Reaction:**

The catalytic performance of the catalysts was evaluated in a fixed bed reactor. 80 mg catalyst was loaded in a quartz-lining stainless steel reactor. The feed gas was a mixture of 32% CO, 64%  $H_2$  and 4% Ar. In a typical reaction, the pressure and gas flow rate was set for 30 bar and 20 mL/min

(GHSV=15000 mL·g<sup>-1</sup>.h<sup>-1</sup>) respectively. Then, the reaction tube was heated from room temperature to 270 °C at 5 °C /min, and the reaction was conducted at 270 °C. It is worth noting that no reduction or carburizing pretreatment was carried out before reaction. The gas phase products were analyzed by an Agilent 6890 GC equipped with an FID and TCD detector, with 4% Ar as inert standard. The heavier hydrocarbons were cooled down and collected in a trap, and analyzed offline by an Agilent GC 7820, with a HP-5 capillary column and FID detector. The products selectivity was calculated on a carbon basis.

#### **STPSR Experiment**

Prior to the STPSR experiments, the Fe catalyst was treated in H2 (20 mL/min) and the Fe5C2 catalyst was treated in 10% C2H4/H2 mixture at 300 oC for 2 hrs to remove the surface contaminates. After cooling down to RT, the gas flow was switched to synthesis gas of 20 mL/min and the pressure was raised to 30 bar. Then, the reactor was heated to 150 °C at 20 °C /min, and kept steady for 2 hrs. Afterward, the temperature was elevated 20 °C higher and held for 2 hrs at that temperature. The process was repeated until the reaction temperature reached 270 °C. A Pfeiffer mass spectrometer (MS) Omnistar was used to analyze on-line the reactants and products. The M/e value detected as follows: 2 for hydrogen, 15 for methane, 18 for water, 26, 27, 30 for  $C_2$  products (acetylene, ethylene, and ethane), 28 for CO, 42 for  $C_3$ , 44 for  $CO_2$ , 56 for  $C_{4+}$  and 70 for  $C_{5+}$ .

#### **Catalyst Characterization**

The TEM experiments were conducted at a FEI Tecnai F30 transmission electron microscope operating at 300 keV. The XPS experiments were carried out in an Axis Ultra imaging photo electron spectrometer with Al kα as X-ray source. The binding energy of graphite carbon was calibrated to 284.8 eV. X-ray powder diffraction data were collected at a Rigaku DMAX-2400 equipped with Cu Al Kα radiation. The Raman characterizations were performed on a Renishaw 1000 Raman imaging microscope system with an excitation wavelength of 632.8 nm. The <sup>57</sup>Fe Mössbauer Effect spectra of

as-synthesized iron carbide NPs were collected by a Topological 500A spectrometer and a proportional counter at room temperature. The  $\gamma$  radiative source was a  $^{57}$ Co (Rh) moving with constant acceleration mode. The XAFS spectra were collected at beam line 14W of Shanghai Synchrotron Facility (SSRF) in transmission mode with Si (111) monochromator. The samples for characterization were prepared and transferred under protection of Ar. As metallic iron and iron carbide were very sensitive for oxidation, sometimes the oxidation of the sample is unavoidable. For TEM study, the used catalyst was passivated in 0.5%  $O_2$  in Ar at room temperature for 1 hr before the measurement.

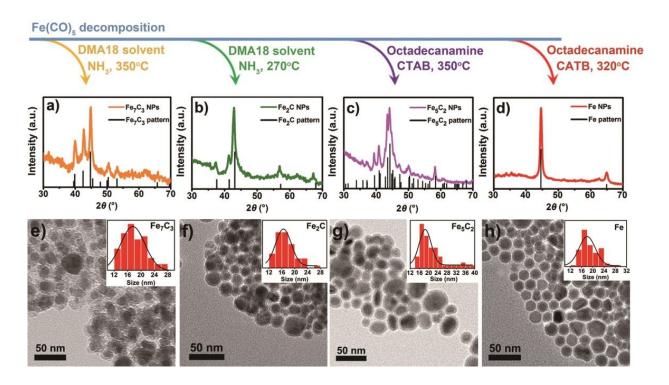
#### Calculation

Spin-polarized DFT calculations have been performed by using Vienna Ab Initio Simulation Package (VASP). $^{46-47}$  Throughout the calculations, projector augmented wave (PAW) potential $^{48}$  and the generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional $^{49}$  was adopted. The planewave cutoff energy was set by 400 eV. The force and energy convergence standards were 0.02 eV/Å and  $1\times10^{-4}$  eV, respectively. Monkhorst-Pack $^{50}$  k-points sampling of  $3\times7\times7$  and  $10\times10\times10$  were adopted for Fe $_5$ C $_2$  and Fe bulk calculations with monoclinic (space group C2/c) and body centered cubic (BCC) crystal structures, respectively. The optimized lattice constant of Fe $_5$ C $_2$  is a=11.55 Å, b=4.50 Å and c=4.99 Å with  $\beta$ =97.6°, which are consistent with experimental findings that a=11.59 Å, b=4.58 Å and c=5.06 Å with  $\beta$ =97.75°. $^{51}$  The determined lattice constant (2.83 Å) for Fe with body centered cubic (BCC) crystal structure also agrees very well with experiment. $^{52}$ 

Fe-terminated  $Fe_5C_2$  (100) surface was simulated by a slab of seven-layered Fe atoms and three-layered C atoms. In the calculations, the topmost four Fe and one C layers were fully relaxed, whereas the remained atoms were fixed in their bulk positions. For BCC-Fe (310) surface, ten Fe atoms layered slab model was used, and only the top five Fe layers are allowed to relax. A p(2×2) unit

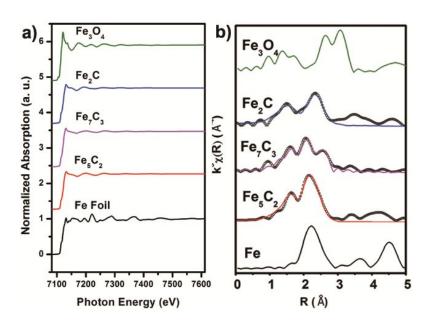
cell was utilized for the two considered slab models. We have used Monkhorst–Pack mesh k-points of  $3\times3\times1$  for Fe<sub>5</sub>C<sub>2</sub> (100), and  $5\times5\times1$  for BCC-Fe (310) surface. The vacuum region along the z direction was specified by 15 Å, and the dipole correction was considered in our calculations. Force-reversed method<sup>53</sup> was used to locate the transition states (TSs) and a force tolerance of 0.03 eV/Å was applied without zero-point correction. Some TSs along the minimum-energy reaction pathways were also reaffirmed by using the climbing image nudged elastic band (CI-NEB) method.<sup>54</sup> For a given elementary reaction, we considered the separately adsorption of intermediates at their most favourable adsorption sites as the initial and final states for the reaction barriers calculations.

### **Results and Discussion**



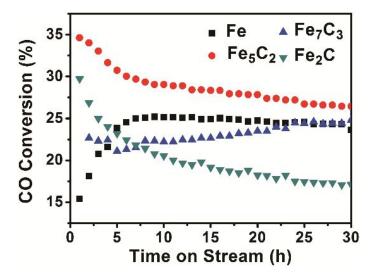
**Fig. 1.** XRD patterns and TEM images of of  $Fe_7C_3(a, e)$ ,  $Fe_2C(b, f)$ ,  $Fe_5C_2(c, g)$  and  $\alpha$ -Fe (d, h) NPs. It is clear from the TEM images and XRD profiles that all the obtained Fe and iron carbide nanoparticles have the pure phase structure and similar size (around 18 nm).

In fabrication of both Fe $_7$ C $_3$  and Fe $_2$ C NPs, NH $_3$  was chosen as atmosphere as well as inducing agent while Fe(CO) $_5$  was used as precursor. In particular, Fe $_7$ C $_3$  NPs were obtained in N,N-dimethyloctadecylamine solvent under 350 °C for 2 h and Fe $_2$ C NPs were produced in dodecylamine solvent under 260 °C for 2 h (see Supporting Information).  $\alpha$ -Fe and Fe $_5$ C $_2$  NPs were synthesized via a bromide-induced process as described elsewhere. Fig. 1 presented the X-ray Powder Diffraction (XRD) patterns and corresponding transition electron microscope (TEM) images of as-synthesized Fe $_7$ C $_3$ , Fe $_2$ C, Fe $_5$ C $_2$  and  $\alpha$ -Fe NPs. According to the XRD results, the peaks in each sample were in good consistence with the standard patterns, which suggests the generation of single phase in each of the four samples. Furthermore, TEM images of the four samples indicate that they are all spherical NPs with size around 18 nm.



**Fig. 2**. a) XANES and b) EXAFS spectra of  $Fe_5C_2$ ,  $Fe_7C_3$  and  $Fe_2C$ . The XANES and EXAFS spectra of Fe foil and standard pattern of  $Fe_3O_4$  are shown by black curve and green curve, respectively. The curves composed by black circles in b) indicate the experimental data of each iron carbide and the colored curves indicate the simulation curves. The XAFS samples were prepared and transferred under protection of Ar.

The X-ray Absorption Fine Structure (XAFS) and Mössbauer spectra data also support this conclusion. Fe K edge XANES suggests that the iron carbide particles exhibit very low oxidation state as compared to the metallic Fe foil. The relatively low frequency oscillation of the post-edge features indicates Fe central atom has neighbours with small bond length. Further EXAFS fitting results confirm that all of the Fe carbide particles synthesized have Fe-C coordination shell near 2.0 Å, the average first Fe-Fe shell bond length expands from 2.46 Å to around 2.60 Å, due to the incorporation of carbon into the BCC lattice of  $\alpha$ -Fe (Fig. 2). Furthermore, no features of Fe oxides and Fe are observed, suggesting all the particles are pure carbide. Moreover, the Mössbauer spectra have been shown in Fig. S4. The sextet peaks indicates the formation of Fe<sub>2</sub>C, Fe<sub>7</sub>C<sub>3</sub> and Fe<sub>5</sub>C<sub>2</sub> NPs, while the weak doublet peaks in Fe<sub>5</sub>C<sub>2</sub> suggests that the Fe<sub>5</sub>C<sub>2</sub> NPs may have better crystallinity compare with Fe<sub>2</sub>C and Fe<sub>7</sub>C<sub>3</sub>. Therefore, single-phase nature of these  $\alpha$ -Fe and iron carbide NPs along with their similar morphology provide us with ideal platforms for the investigation of their intrinsic catalytic behavior and structural evolution in FTS process. At the same time, as the cementite (Fe<sub>3</sub>C) was reported to have poor activity in FTS,  $^{30,32}$  it is not discussed in this work.



**Fig. 3.** CO conversion as a function of time over Fe, Fe<sub>2</sub>C, Fe<sub>7</sub>C<sub>3</sub> and Fe<sub>5</sub>C<sub>2</sub> catalysts (reaction conditions: 270 °C, 30 bar, 20 ml/min syngas).

The  $\alpha$ -Fe and iron carbide NPs were dispersed on silica support and directly used in FTS reaction (3 MPa syngas, 270 °C). For a 40 h reaction, the product distribution on Fe and Fe<sub>5</sub>C<sub>2</sub> are similar except Fe has a higher C<sub>5+</sub> selectivity (49%). Fe<sub>2</sub>C has the highest selectivity towards CO<sub>2</sub> (22.7%), while Fe<sub>2</sub>C and Fe<sub>7</sub>C<sub>3</sub> show considerably high selectivity of 20.1% and 19.5% toward methane, respectively (Fig. S5). Both CO<sub>2</sub> and CH<sub>4</sub> are undesired products. CO conversion on prepared catalysts with time on stream is shown in Fig. 3 with FTS activity shown in Table 1.

**Table 1.** The surface-specific activity (TOF) and metal-mass-based activity (Activity) excluded CO<sub>2</sub> formation over various iron and iron carbides catalysts.

Catalysts	Temperatur e	Pressure	TOS	TOF <sub>FTS</sub>	Activity	Reference
Calalysis	(°C)	(bar)	(h)	(s <sup>-1</sup> )	(1×10 <sup>-4</sup> Mol <sub>CO</sub> ·g <sub>Fe</sub> <sup>-1</sup> ·s <sup>-1</sup> )	Reference
Fe	270	30	1	0.13	1.1	This work
Fe	270	30	30	0.20	1.7	This work
Fe <sub>5</sub> C <sub>2</sub>	270	30	1	0.29	2.3	This work
Fe <sub>5</sub> C <sub>2</sub>	270	30	30	0.22	1.8	This work
Fe <sub>7</sub> C <sub>3</sub>	270	30	1	0.16	1.6	This work
Fe <sub>7</sub> C <sub>3</sub>	270	30	30	0.18	1.7	This work
Fe <sub>2</sub> C	270	30	1	0.16	1.8	This work
Fe <sub>2</sub> C	270	30	30	0.09	1.0	This work
Fe <sub>x</sub> O <sub>y</sub> @C	270	20	-	-	0.31	55
Fe/SiO <sub>2</sub>	270	20	-	-	0.20	56
Fe-in-CNT	270	50	-	-	2.5	57
25-Fe@C	340	20	-	0.11	4.9	58
RQ Fe	200	30	-	-	3.5	59

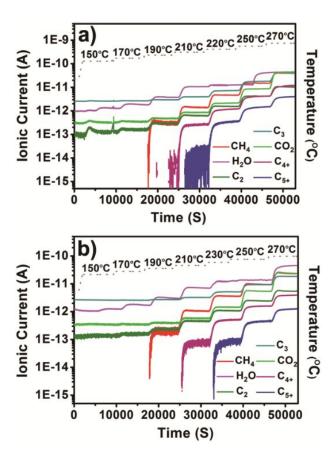


Fig. 4. STPSR profile on Fe<sub>5</sub>C<sub>2</sub> (a) and Fe (b) catalysts.

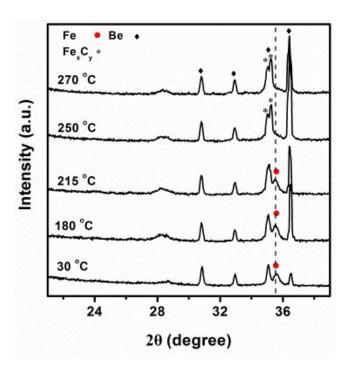
At the beginning,  $Fe_5C_2$  exhibites the highest CO conversion (around 35%), followed by  $Fe_2C$  (29.5%) and  $Fe_7C_3$  (22.5%), while  $\alpha$ -Fe had the lowest CO conversion (15.2%). The initial CO reactivity reflects the intrinsic catalytic properties of  $Fe_2C$ ,  $Fe_7C_3$  and  $Fe_5C_2$ . With the reaction proceeding, the activities of the four catalysts showed different trends. For  $Fe_5C_2$  and  $Fe_2C$ , the CO conversion decreased sharply in the first 6 hrs but gradually stabilized later on. For  $Fe_7C_3$ , CO conversion dropped slightly in the first 5 hrs, after a small increase, became stabilized. On the contrary, that of  $\alpha$ -Fe catalyst increased rapidly in the first 7 hrs, and then remained almost constant afterwards. When the conversion for all catalysts approached the steady state,  $Fe_5C_2$  remains the most active one with a conversion of 27%, and the activities of  $Fe_7C_3$  and  $Fe_7C_3$  and

 $x10^{-4}$  Mol<sub>CO·gFe-1</sub>·s·1). After reaction, the morphologies of used Fe<sub>5</sub>C<sub>2</sub> and Fe<sub>2</sub>C catalysts were maintained with main phases transformed into a mixture of iron oxide and original carbide, whereas for Fe<sub>7</sub>C<sub>3</sub>, the formation of Fe<sub>5</sub>C<sub>2</sub> was observed (Fig. S6 and S7). Significantly, the main phase of used  $\alpha$ -Fe catalyst was transformed into Fe<sub>5</sub>C<sub>2</sub>, in addition to Fe<sub>3</sub>O<sub>4</sub> (Fig. S6). There has always been a debate on whether and why metal and/or carbide is the active phase for FTS. To examine this, we chose Fe<sub>5</sub>C<sub>2</sub> as a representative for carbide phase and compared its catalytic behavior in the early stage of FTS with that of  $\alpha$ -Fe catalyst. We designed a high-pressure stepwise temperature programming surface reaction (STPSR) apparatus, to allow the observation of reaction kinetics at high pressure. This TPSR experiment enables the observation of those masked information by steady-state reaction evaluation.<sup>17, 60-61</sup>

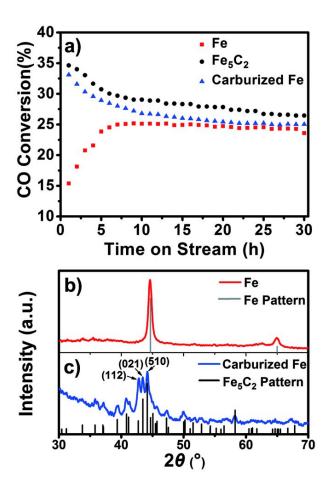
The formation of various products during the STPRS of syngas over the  $Fe_5C_2$  catalyst is illustrated in Fig. 4a. Before 150 °C, no products were formed, which indicates that syngas cannot be activated over  $Fe_5C_2$  at temperatures below 150 °C. With the reaction temperature reaching 150 °C and remaining for 20 min, the formation of water,  $C_2$  hydrocarbons (mostly acetylene and ethylene) and  $CO_2$  were observed. When the temperature was raised to 170 °C, with the appearance of  $C_3$  hydrocarbon, the amount of water and  $C_2$  formed were considerably high. Yet, to our surprise, methane was not detected until then. The MS signal for methane appeared only when the reaction temperature reached 190 °C. The formation of methane at higher temperature than that of  $C_2$  hydrocarbons indicates that the hydrogenation of monomers toward methane is kinetically less favorable than C-C coupling. When the temperature reached 210 °C,  $C_4$  compounds appear, followed with the formation of  $C_{5+}$  hydrocarbons until 20 min later.

To our surprise, STPSR profile of  $\alpha$ -Fe catalyst shows a distinct behavior as compared with that of Fe<sub>5</sub>C<sub>2</sub> (Fig. 4b). At 150 °C, water formation was reproducibly observed, whereas no C<sub>2</sub> hydrocarbons could be detected in 2 hrs. Instead, C<sub>2</sub> hydrocarbon appeared only at around 10 min after the temperature reaching 170 °C. The formation temperature of methane, C<sub>3</sub> alkane, and CO<sub>2</sub>

was at least 190 °C.  $C_4$  and  $C_5$  formed at 210 °C and 230 °C, which are very similar to those of  $Fe_5C_2$ . Thus, although  $C_2$  hydrocarbon on  $\alpha$ -Fe formed at a higher temperature (170 °C) than that on  $Fe_5C_2$  (150 °C), the formation of methane on both catalysts requires a temperature of 190 °C or higher. Namely, methane formation on both catalysts remains less favorable, compared with the formation of water and  $C_2$  hydrocarbon. The formation of water in STPSR process is interesting as the preadsorbed water on the catalysts was removed through a pretreatment at 300°C before STPSR. Therefore, the water can only be formed from the reaction between hydrogen and oxygen adsorbed on the catalyst. There are two possible sources for adsorbed oxygen, namely, either the product from CO dissociation, or contaminated molecular oxygen from gas phase that was not removed by hydrogen pretreatment before STPSR. In fact, the latter oxygen source can be excluded by a  $H_2$ -STPSR experiment (Fig. S8). This implies that the oxygen source of water on  $\alpha$ -Fe catalyst can only be the dissociated O from CO. Therefore, it is concluded that CO dissociation took place at 150°C over Fe catalyst, which is also the case for Fe $_5C_2$  catalyst.



**Fig. 5.** In-situ XRD patterns of Fe NPs treated with 2 MPa syngas at various temperatures. (XRD was recorded in beam line 14B of Shanghai Synchrotron Facility. The diamonds mark the diffractions from beryllium-made in-situ cell).



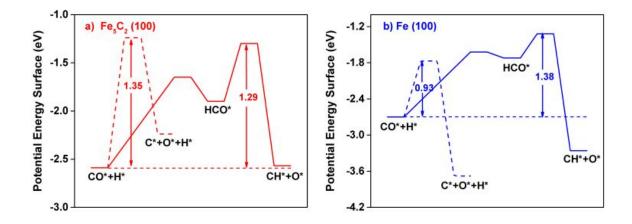
**Fig. 6.** (a) CO conversion of α-Fe, Fe<sub>5</sub>C<sub>2</sub> and pre-carburized α-Fe NPs catalysts. XRD pattern of α-Fe NPs catalysts (b) before and (c) after carburization. It is clear that carburizing the supported α-Fe catalyst with ethylene or the mixture of ethylene and hydrogen at 350 °C for 1 hr would transform α-Fe to Fe<sub>5</sub>C<sub>2</sub>.

Dissociated carbon would be hydrogenated, forming subsequently the hydrocarbons. 62-63 At 150 °C, the formation of the hydrocarbons was observed on Fe<sub>5</sub>C<sub>2</sub>, but not on  $\alpha$ -Fe within the first 2 hrs. This implies that dissociated carbon would accumulate on α-Fe, and might diffuse into the interstitial sites of subsurface and bulk  $\alpha$ -Fe region, carburizing  $\alpha$ -Fe toward carbide. Fe nanoparticles can be easily carburized into iron carbides, 64 whereas the transformation from the single crystals and sheets of Fe to iron carbide is difficult<sup>65-66</sup> which can be attributed to the difficulty for carbon atoms permeating into the interstitial sites formed by the close packing of Fe atoms. Thermodynamically, this is possible since the corresponding Gibbs free formation energy (-3.22 eV per chemical formula unit of Fe<sub>5</sub>C<sub>2</sub> under FTS conditions) at this condition is exothermic (Section 8 in SI). The above hypothesis was confirmed by in-situ XRD experiments under FTS reaction condition (2 MPa syngas, Fig. 5). It can be found that under syngas stream, α-Fe, which corresponds to a diffraction with two theta value of 35.3°, was gradually transformed to iron carbides phase (two theta value of 34.8°, most like Fe<sub>5</sub>C<sub>2</sub>) when increasing the reaction temperature to that higher than 220 °C. Although the reaction condition is not exactly same with that in STPSR experiments, this tells compellingly that in FTS process, α-Fe is indeed apt to react with surface carbon from dissociated CO, forming iron carbide. Moreover, when we carburized the supported α-Fe catalyst with ethylene or the mixture of ethylene and hydrogen to get supported iron carbide catalyst (Fe<sub>5</sub>C<sub>2</sub>, as evidenced by XRD, Fig. 6b and c), we observed that the initial activity of the resulted carburized catalyst increased dramatically (Fig. 6a), reaching 33%, very close to the initial activity of pure phase Fe<sub>5</sub>C<sub>2</sub> catalyst. The activity evolution of the carburized Fe catalyst almost duplicated that of Fe<sub>5</sub>C<sub>2</sub> catalyst, i.e. it drops gradually in the first few hrs and becomes relatively stable after around 15 h reaction. The induction period observed on α-Fe catalyst could therefore be attributed to the process of carburization. Once the carburization was completed and Fe<sub>5</sub>C<sub>2</sub> was formed, the resulted catalysts would show a higher activity because of its higher intrinsic activity of Fe<sub>5</sub>C<sub>2</sub>.

For the drop of activity of  $Fe_5C_2$  catalyst with reaction time (Fig. 3), it was attributed to the oxidation of iron carbide by the oxidative products such as  $H_2O$  and  $CO_2$  to the inactive iron oxide.

This was confirmed by the activity measurement of pure phase  $Fe_5C_2$  treated with  $CO_2$  at different temperature (See Fig. S9). Indeed, both the initial crystal phase of the catalyst and the reaction atmosphere could affect the structural evolution and eventually equilibrium structure of iron-based catalysts under reaction condition.

To rationalize the experimental results presented above, it is essential to study the intrinsic FTS activity of the pristine iron, and the difference with the iron carbide. Herein, DFT calculations were performed to study the crucial FTS steps including CO activation, C-C coupling and methane formation. The stepped Fe (310) and Fe-terminated Fe<sub>5</sub>C<sub>2</sub> (100) surfaces (Fig. S10) were used to model the corresponding iron and iron carbides catalysts, respectively. The stepped Fe (310) was chosen here because Fe(310) surface occupies a large surface area proportion of iron Wulff shape by 22%<sup>67</sup> and it shows higher activity for CO dissociation as compared with the traditional (110), (100), (211) and (111) surfaces. <sup>68</sup> Fe<sub>5</sub>C<sub>2</sub>(100) was selected here since it is one of the largest exposed surfaces under operating FTS conditions (600 K, 10 bar, H<sub>2</sub>/CO = 2.5) on Fe<sub>5</sub>C<sub>2</sub> Wulff construction.<sup>69</sup> On C-terminated Fe<sub>5</sub>C<sub>2</sub> surface, there will be no active sites for CO activation or the activity of CO activation will be significant low. 70-72 Herein, in the present work, we have adopted Fe-terminated Fe<sub>5</sub>C<sub>2</sub>(100) surface which stands for the C-terminated  $Fe_5C_2$  (100) surface with abundant C vacancies, where CO dissociation is feasible.<sup>39, 70</sup> Additionally, Fe(310) and Fe-terminated Fe<sub>5</sub>C<sub>2</sub> (100) surfaces have the same B5 step surface and it is feasible for the direct activity and selectivity comparison between Fe and Fe<sub>5</sub>C<sub>2</sub>.<sup>73-74</sup> The calculated binding energetics of the important intermediates CO\*, C\* and CH\* at their most favorable sites were -2.07, -8.15 and -6.90 eV on Fe(310) while -1.91, -7.08 and -6.53 eV on Fe<sub>5</sub>C<sub>2</sub>(100) (Table S2), respectively. As expected, pristine iron is more reactive than iron carbide, and the binding strength toward C\* is 1.07 eV stronger. This is determinative to the distinct activities of metal and carbide phases on CO activation, C-C coupling and methane formation.



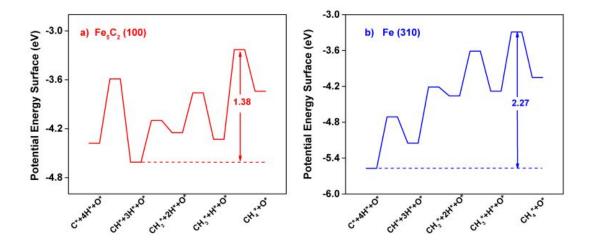
**Fig. 7**. The potential energy diagrams for CO dissociation on (a)  $Fe_5C_2$  (100) (red) and (b) Fe (310) (blue) surfaces. The solid and dashed lines present direct and H-assisted CO activation pathways, respectively. The apparent activation barriers (in eV) are indicated. The corresponding geometries involved in CO activation are shown in Fig. S11 and Fig. S12.

For Fe(310), calculated  $E_{\rm app}$  for CO dissociation was 0.93 eV, with an exothermic reaction energy (-0.98 eV) due to the strong C-Fe binding (Fig. 7a). The small barrier is in good accordance with observation of water formation at 150 °C. For C-C coupling, there are many possible pathways between CH<sub>i</sub> and CH<sub>j</sub> (i, j = 0 ~ 3). To evaluate their relative activity, equilibrium concentration of the corresponding monomers, which is proportional to their formation energy  $E_{\rm f}$  (with zero energy reference of CO dissociation product), should be considered. We define accordingly apparent barrier  $E_{\rm app}$ , namely, summation of formation energy  $E_{\rm f}$  of CH<sub>i</sub> and CH<sub>j</sub> and their coupling barrier  $E_{\rm c}$ . It is found that the strong C-Fe binding not only makes the formation of CH<sub>i</sub> and CH<sub>j</sub> monomers energetically highly cost with  $E_{\rm f}$  falling in a magnitude of 2.49 eV, but also leads CH<sub>i</sub>-CH<sub>j</sub> coupling kinetically very demanding with maximum  $E_{\rm c}$  of 1.99 eV (Table 2 and Fig. S15). Indeed, among all possible CH<sub>i</sub>-CH<sub>j</sub> coupling considered, the least  $E_{\rm app}$  (=  $E_{\rm f}$  +  $E_{\rm c}$ ) calculated was as large as 1.87 eV from C\*-CH\* coupling. Methane formation was also found highly demanding with an overall barrier of 2.27 eV (Table S3 and Fig. 8).

**Table 2**. Formation energy ( $E_f$ ) of  $CH_x^*$  and  $CH_y^*$ , elementary coupling barrier ( $E_c$ ) between  $CH_x^*$  and  $CH_y^*$ , and the apparent barrier ( $E_{app} = E_f + E_c$ ) on  $Fe_5C_2$  (100) and Fe (310) surfaces. The formation energy was calculated with respect to  $CH^*$  on  $Fe_5C_2$  (100) surface and the atomic carbon  $C^*$  on the Fe (310) surface since they are located at the lowest valley of the whole potential surface.

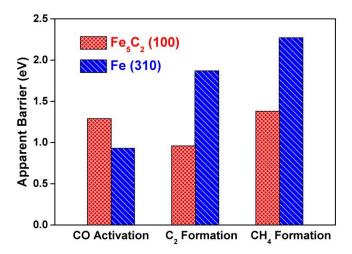
Reaction	Fe <sub>5</sub> C <sub>2</sub> (10	00)		Fe (310)	Fe (310)		
	E <sub>c</sub> (eV)	E <sub>f</sub> (eV)	E <sub>app</sub> (eV)	E <sub>c</sub> (eV)	E <sub>f</sub> (eV)	E <sub>app</sub> (eV)	
C*-C*	0.84	0.47	1.31	1.99	0.00	1.99	
C*-CH*	0.78	0.24	1.02	1.45	0.42	1.87	
C*-CH <sub>2</sub> *	0.77	0.59	1.36	1.16	1.20	2.36	
C*-CH <sub>3</sub> *	1.13	0.52	1.65	1.03	1.29	2.32	
CH*-CH*	1.02	0.00	1.02	1.38	0.85	2.23	
CH*-CH <sub>2</sub> *	0.60	0.36	0.96	1.24	1.63	2.87	
CH*-CH <sub>3</sub> *	1.19	0.28	1.47	1.35	1.71	3.06	
CH <sub>2</sub> *-CH <sub>2</sub> *	0.48	0.72	1.20	0.79	2.41	3.20	
CH <sub>2</sub> *-CH <sub>3</sub> *	1.16	0.64	1.80	1.07	2.49	3.56	

The large barriers for C-C coupling and methane formation on Fe(310) leads the pristine iron a rather poor FTS activity. This DFT calculations are also consistent with previous experiments resuts that iron catalysts contain no carbide have the lowest FTS activity. The dissociated carbon from CO would therefore stay sufficient long time on catalyst surface, which is necessary for carburization and transition toward thermodynamically more favorable carbide. As a result, the unsaturated C<sub>2</sub> hydrocarbon observed on iron at 170 °C (Fig. 4b) cannot come from the pristine iron. Instead, it should come from the carburized iron catalysts, namely, iron carbide. Since dissociated carbon was already available at 150 °C on the pristine iron, this also means the carburization process requires the activation temperature of 170 °C at least.



**Fig. 8.** The potential energy diagram for methane formation on (a) Fe<sub>5</sub>C<sub>2</sub> (100) (red) and (b) Fe (310) (blue) surfaces. The apparent activation barriers (in eV) are indicated. The corresponding transition state geometries are shown in Fig. S14.

For Fe<sub>3</sub>C<sub>2</sub>(100), the calculated  $E_{app}$  for CO dissociation is 1.29 eV with an endothermic reaction energy (Fig. 7b), due to the destabilization of dissociated C\* on carbide surface. Though CO dissociation barrier is higher than that of Fe(310), its modest value makes CO dissociation on Fe<sub>5</sub>C<sub>2</sub>(100) remain facile, for instance at 150 °C. This is still in accordance with experimental observation of water formation at this temperature. On the other hand, the destabilization of surface C\* and CH\* on carbide (Table S2) promotes greatly the C-C coupling and methane formation. The formation of CH<sub>i</sub> and CH<sub>j</sub> monomers become energetically much less cost with  $E_f$  falling in range of 0 ~ 0.72 eV, and the CH<sub>i</sub>-CH<sub>j</sub> coupling becomes kinetically facile too with maximum  $E_c$  of 1.19 eV (Table 2 and Fig. S16). As a result, the least  $E_{app}$  calculated for the C-C coupling is only 0.96 eV from CH\*-CH<sub>2</sub>\* coupling. Moreover, C\*-CH\*, CH\*-CH\*, and CH<sub>2</sub>\*-CH<sub>2</sub>\* coupling are also kinetically favorable with  $E_{app}$  of 1.02, 1.02 and 1.20 eV, respectively. For methane formation, the overall barrier also decreases significantly to 1.38 eV (Table S4 and Fig. 8).



**Fig. 9.** Apparent barriers comparison for CO activation, C<sub>2</sub> and CH<sub>4</sub> formation on Fe<sub>5</sub>C<sub>2</sub> (100) and Fe (310) surfaces.

As seen the apparent reaction barriers from Fig. 9, compared to CO activation on Fe $_5$ C $_2$  (100) with barrier of 1.29 eV, the C-C coupling barriers are even lower. This means once CO dissociates on Fe $_5$ C $_2$ , all these C-C coupling pathways could take place right away. This corroborates nicely the experiment over Fe $_5$ C $_2$  at 150 °C (Fig. 4a), namely, once syngas were activated, both acetylene and ethylene were observed simultaneously with formation of water. Meanwhile, the barrier for methane formation remains higher than that of CO activation, a fact of that also found on Fe (310). This implies that a higher temperature for methane formation would be required, which was indeed found in corresponding STPSR experiment (Fig.4a).

#### Conclusion

Through a modified liquid-phase route, a series of pure-phase metallic iron and iron carbide NPs were successfully synthesized, which provided an ideal platform to investigate the fundamentals in iron-based FTS reaction that was not revealed previously. Fe<sub>2</sub>C, Fe<sub>7</sub>C<sub>3</sub> and Fe<sub>5</sub>C<sub>2</sub> were all found to be active for FTS reaction, with Fe<sub>5</sub>C<sub>2</sub> the most active phase. By using a house-designed STPSR method, transient information about the surface species and their reactivity were revealed. DFT calculations showed that iron carbide is intrinsically more active than the pristine iron for C-C coupling and methane

formation because of the strong binding of dissociated atomic carbon on iron. Furthermore, C-C coupling was easier compared with methane formation on iron carbide, which makes iron carbide highly active for FTS with good olefin selectivity. The distinct activities of different iron phases (metal versus carbide) revealed as well as their stability under FTS condition could be used to design more efficient iron-based FTS catalysts.

# **Supporting Information**

Indicate that Supporting Information is included using the following statement: Supporting Information is available and includes the XRD Spectra (Fig. S1 – S2); physical absorption characterization of the support material (Fig. S3); Mössbauer effect spectra (Table S1 and Fig. S4); products distribution of catalysts (Fig. S5); TEM images and XRD patterns of used catalysts (Fig. S6 – S7); TPSR of Fe and  $Fe_5C_2$  in pure  $H_2$  (Fig. S8); XAFS spectra (Fig. S9) and FTS performances of  $CO_2$  treated  $Fe_5C_2$  catalyst; energetic and geometric information for CO activation,  $CH_4$  formation and  $CH_x$ - $CH_x$  coupling by DFT calculations (Table S2 – S4 and Fig. S10 – S16).

# Conflict of Interest (required)

There is no conflict of interest to report.

# **Funding Information (required)**

This work was financially supported by the Natural Science Foundation of China (21725301, 91645115, 21821004, 21932002, 51631001, 91645202, 91945302), the National Key R&D Program of China (2017YFB0602200, 2017YFB0602205, 2018YFA0208603), Natural Science Foundation of Beijing Municipality (2191001) and the Chinese Academy of Sciences Key Project (QYZDJ-SSW-SLH054). XAS experiments were conducted at the Shanghai Synchrotron Radiation Facility (SSRF) and Beijing Synchrotron Radiation Facility (BSRF). XRD experiments were conducted at SSRF. The Super Computing Center of USTC is gratefully acknowledged.

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# **Table of Contents Graphic (required)**

