



# Reduced graphene oxide film modified by tannic acid for high areal performance supercapacitors

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## Abstract

When graphene oxide (GO) was reduced, the stacking of reduced graphene oxide (rGO) sheets would lead to far lower of its specific capacitance than the theoretical value of graphene. In order to solve this problem, we use tannic acid (TA) to modify the rGO layered film by vacuum filtration of the mixture of GO and TA in solution, and then mild thermal reduction at 180°C. Due to the rich redox active functional groups of TA, the introduction of TA can not only alleviate the stacking of rGO sheets and promote the reduction process of GO at relatively low temperature, but also provide additional pseudocapacitance. When used for two-electrode symmetrical supercapacitor in 6 M KOH electrolyte, the TrGO-0.5 gives areal capacitance of 525 mF cm<sup>-2</sup>, and energy density of 72.2 uWh cm<sup>-2</sup> at power density of 250.9 uW cm<sup>-2</sup>. It also has capacitance retention of 91.7% after 10,000 charging/discharging cycles at current density of 4 mA cm<sup>-2</sup>. The TrGO-0.5 based button cell with 2 M 1-ethyl-3-methylimidazole tetrafluoroborate (EMIMBF<sub>4</sub>) as electrolyte shows the practical application to light up three LEDs.

**Keywords** Reduced graphene oxide · Tannic acid · Areal capacitance · Cycling stability · Supercapacitor

## Introduction

Electrochemical energy storage devices become more and more important in modern society, many devices, such as new energy electric vehicles and portable wireless network products, depend on them for energy support [1–3]. Supercapacitors are electrochemical energy storage devices, compared to batteries, supercapacitors have higher power density, shorter charging time, and longer cycle life, but lower energy density [4–6]. Based on the electrode materials

and their energy storage mechanisms, supercapacitors can be divided into electric double layer (EDL) capacitors and pseudocapacitors [7, 8]. Over the past few decades, researchers have focused on designing and fabricating different kinds of electrode materials for supercapacitors. Carbon materials, such as activated carbon [9, 10], carbon nanotubes [11, 12] and graphene [13, 14], are usually used as electrodes in EDL capacitors, and they store energy by accumulating charges at the interfaces of electrode and electrolyte. Conducting polymers (polyaniline [15, 16], polypyrrole [17, 18], polyacetylene [19, 20], etc.) and transition metal oxides (MnO<sub>2</sub> [21], RuO<sub>2</sub> [22], Fe<sub>2</sub>O<sub>3</sub> [23], etc.) are the typical electrode materials for pseudocapacitors, with Faradaic reaction as the underlying mechanism [24]. Therefore, EDL capacitors generally exhibit better rate capability and cycling stability, but pseudocapacitors have relatively higher specific capacitance. Due to its excellent conductivity and high theoretical surface area (2630 m<sup>2</sup> g<sup>-1</sup>), graphene is the ideal electrode material for EDL capacitors [25, 26]. However, the  $\pi$ - $\pi$  interaction between graphene sheets leads to the stacking of rGO sheets during the reduction process from GO to rGO [27], resulting in the inability to fully utilize their surface area. Introducing intercalator materials between graphene sheets, such as transition metal oxides [28, 29], conducting polymers

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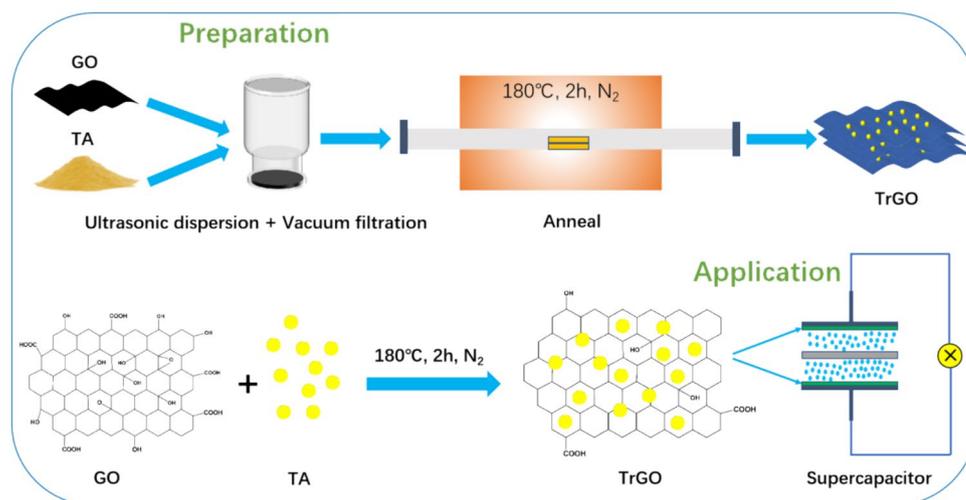
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**Fig. 1** Schematic illustration of fabricating the TrGO film and its application for supercapacitor



[30, 31], and other carbon materials [32–34], as functional components can prevent the stacking of graphene sheets. The synergistic effect between graphene and the intercalator materials makes the graphene-based composites to give better electrochemical performance.

Tannic acid (TA) is an organic molecule which can exhibit redox activity. It mainly consists of derivatives of gallic acid, and sources from various plants such as oak. Because TA is safe, non-toxic, and cost-effective, therefore it is a kind of excellent renewable material. In addition, due to its antioxidant activity, TA can facilitate the reduction process of GO, and serve as cross-linker to capture graphene sheets during the reduction process [35–37]. Therefore, TA is considered as the environmentally friend reducing agent and stabilizer in graphene synthesis [38]. Researchers have reported the application of TA functionalized rGO in biological monitoring [39, 40]. TA was also used to enhance the mechanical property of rGO electrodes for flexible supercapacitors, but the obtained devices have relatively low area capacitance ( $238.74 \text{ mF cm}^{-2}$ ) and energy density ( $3.6 \text{ uWh cm}^{-2}$ ) [41–43].

In this study, TA was used as intercalator, cross-linker and reducing agent during the reduction process from GO to rGO at relatively low temperature of  $180^\circ\text{C}$ , resulting in fabricating TA/rGO (TrGO) film electrodes for high areal performance supercapacitors. The varying content percentage of TA to GO resulted in the change of electrochemical performance of TrGO. When tested in two-electrode symmetrical supercapacitor with 6 M KOH as electrolyte, TrGO-0.5 exhibited an areal capacitance of  $525 \text{ mF cm}^{-2}$ , with capacitance retention of 91.7% after 10,000 charging/discharging cycles. TrGO-0.5 was also assembled into button cell in 2 M EMIMBF<sub>4</sub> electrolyte, which shows practical application to light up three LEDs. This research introduces a novel approach to modify rGO film for electrochemical

energy storage, resulting in improving the potential of graphene-based electrodes for practical applications.

## Experimental sections

### Materials

All materials were used directly after purchase. Graphite powder (325 mesh), tannic acid, hydrogen peroxide ( $\text{H}_2\text{O}_2$ , 30%), potassium permanganate ( $\text{KMnO}_4$ , 99.5%), sulfuric acid ( $\text{H}_2\text{SO}_4$ , 95–98%), hydrochloric acid (HCl, 37–38 wt%), potassium hydroxide (KOH), dimethylpyrrolidone (DMF) were bought from Shanghai Sinopsin Chemical Reagent Co., LTD. Polytetrafluoroethylene filter membrane (aperture:  $\sim 0.45 \mu\text{m}$ ) and N, N-Dimethylformamide (EMIMBF<sub>4</sub>) were bought from Jinlong Trading Co., LTD. and Shanghai YinYe Investment Co., LTD., respectively.

### Preparation of TrGO electrodes

GO was synthesized via the improved Hummers method, the detailed process was described in supporting materials. To prepare TA/GO films, taking 50 mg of GO, pouring it into 20 ml of DMF. After ultrasonating for 1 h to obtain GO dispersion, mixing GO with TA at different GO: TA mass ratios (1:0.25; 1:0.5, 1:1, and 1:2) for 30 min under stirring. After ultrasonic dispersion for another 30 min, pouring the mixture into a filter bottle to vacuum filtrate, then air-dry naturally to obtain TA/GO films. The TA/GO films were annealed in a horizontal tube furnace at  $180^\circ\text{C}$  for 2 h (0.5 MPa,  $\text{N}_2$  environment) to obtain TrGO films, named as rGO, TrGO-0.25, TrGO-0.5, TrGO-1 and TrGO-2. The fabrication process of TrGO films was illustrated in Fig. 1.

## Materials characterization

Scanning electron microscope (SEM, Gemini 500 nCarl Zeiss, Germany) was used to observe the surface and cross-section morphologies of the samples. Raman spectrometer (LabRAM HR Evolution) was employed for microspectroscopy in the range of  $500\sim 2000\text{ cm}^{-1}$ , with an excitation wavelength of 785 nm. X-ray diffraction (XRD) spectra were obtained in the  $5\sim 70^\circ$  diffraction angle range using PANalytical X-PERT PRO MPD. X-ray photoelectron spectroscopy was analyzed for the surface properties of the electrodes by using an X-ray photoelectron spectrometer (XPS, ESCALAB250Xi).

## Electrochemical characterization

The two-electrode cells with 6 M KOH as electrolyte were built to test the electrochemical performance of the samples (Fig. 1). The masses of different electrodes are listed in Table S1. Cyclic voltammetry (CV), Galvanostatic charge-discharge (GCD) and Electrochemical impedance spectroscopy (EIS) measurements of the samples were tested by the electrochemical workstation (CHI660E, Chenhua Instrument, China). The scan rates of CV range from 10 to  $500\text{ mV s}^{-1}$  within the voltage window from 0 to 1 V. GCD tests were conducted at current densities from  $0.5$  to  $8\text{ mA cm}^{-2}$  within the potential window from 0 to 1 V. EIS was performed over frequency range from 1 to 1 MHz with 5 mV amplitude. Button cells were also assembled in 2 M EMIMBF<sub>4</sub> electrolyte to test the electrochemical performance of TrGO-0.5 via CV (scan rates from 10 to  $500\text{ mV s}^{-1}$  with voltages from 0 to 3 V), GCD (current densities from  $0.5$  to  $8\text{ mA cm}^{-2}$  with voltages from 0 to 3 V) and EIS measurements. The cycling stability of TrGO-0.5 was determined by continuous charge-discharge performance at  $0.5\text{ mA cm}^{-2}$  for 10,000 cycles. The specific capacitance (C), energy density (E), and power density (P) were calculated based on GCD discharge curves via the equations provided in supporting materials.

## Results and discussion

### Material characterizations

The surface and cross-section morphologies of the samples were observed via SEM (Fig. 2 and Fig. S1 and S2). Figure 2a illustrates the surface morphology of rGO film, which reveals a typical rippled and folded structure. In Fig. 2b, the planar morphology of TrGO-0.5 film shows the main characteristics of rGO, with additional particles of TA on rGO surface. These TA particles would contribute to the repair of rGO sheets, and create more folds between rGO

sheets. Figure 2c and d display the surface morphology of TrGO-2, which shows the accumulation of TA particles on rGO surface due to the excess of TA. This abundance of TA particles may potentially lead to the reduction of TrGO-2 electrochemical performance, as further experiments would confirm. Figure 2e presents the cross-sectional view of rGO electrode, revealing relatively dense arrangement of the rGO layers. However, the SEM of TrGO-0.5 electrode cross-section shows a slight expansion between rGO layers, indicating that TA can alleviate the stacking of rGO layers (Fig. 2f).

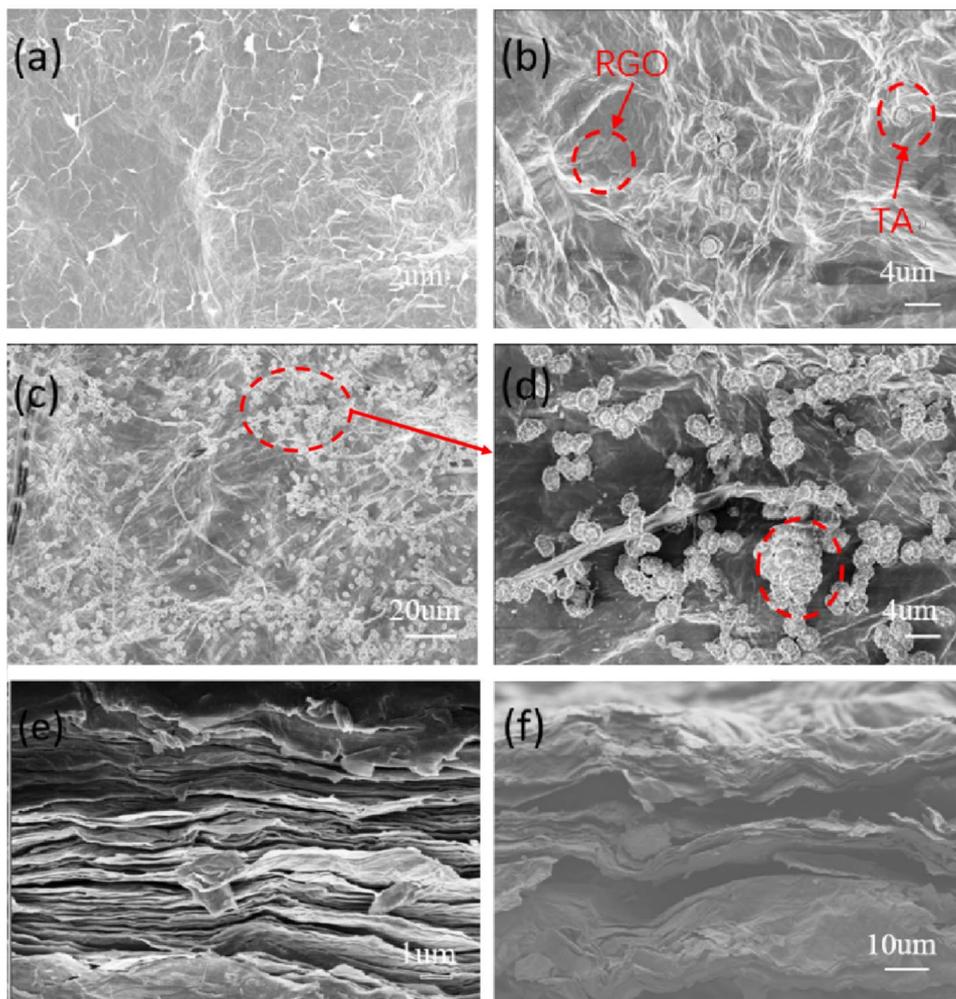
Figure 3a displays the XRD patterns of rGO and TrGO samples. The XRD pattern of rGO shows two broad diffraction peaks which correspond to the (002) and (001) crystal planes of graphene, respectively [44, 45]. However, the diffraction peaks of (002) become sharp for the TrGO samples, which indicates that TA can facilitate the reduction process of GO at relatively low temperature of  $180^\circ\text{C}$ . The Raman spectra of rGO and TrGO samples are shown in Fig. 3b. All samples exhibit two peaks at around  $\sim 1358\text{ cm}^{-1}$  and  $\sim 1597\text{ cm}^{-1}$ , corresponding to the D-band and G-band of graphene, respectively. The D-band represents a mixed vibration mode of  $\text{sp}^3$  defects, while the G-band indicates in-plane vibrations of  $\text{sp}^2$  carbon atoms. The ID/IG ratio is always used to characterize the state of disorder or defects in carbon materials [44–46]. Figure 3c shows that the values of ID/IG of the samples first increase from 1.23 (rGO) to 1.41 (TrGO-1), and then decrease to 1.15 (TrGO-2), indicating that the introduction of TA to rGO not only plays a certain reducing effect but also generates some defects. However, it is not the case that excessive introduction of TA can achieve the best results.

The surface elemental composition and chemical bonding states of the samples were examined by XPS. As shown in Fig. 3d, the XPS spectra of rGO and TrGO-0.5 indicate that they are mainly composed of C1s and O1s (The carbon and oxygen contents are listed in Table S2). The three peaks at about 284.5, 286, and 288.4 eV can be assigned to C-C/C=C, C-O, and O=C-O chemical bonds, respectively (Fig. 3e and f). It is noteworthy that the intensity of C-O peak of TrGO-0.5 is higher than that of rGO, caused by the rich oxygen-containing groups of TA (such as phenolic hydroxyl and carboxyl groups) in TrGO-0.5. The oxygen content of TrGO-0.5 is 17.23 at%, and it is 15.08 at% for rGO (Table S2). The higher oxygen content of TrGO-0.5 also indicates that the main functional groups of TA still exist after mild thermal reduction process of TA/GO film at  $180^\circ\text{C}$ .

### Electrochemical performance

The electrochemical performances of the samples were tested via two-electrode symmetric cell in 6 M KOH

**Fig. 2** The surface and cross-section morphologies of samples. **a, b** The SEM images of rGO and TrGO-0.5. **c, d** The SEM images of TrGO-2 at different magnification scale. **e, f** The cross-sectional SEM images of rGO and TrGO-0.5

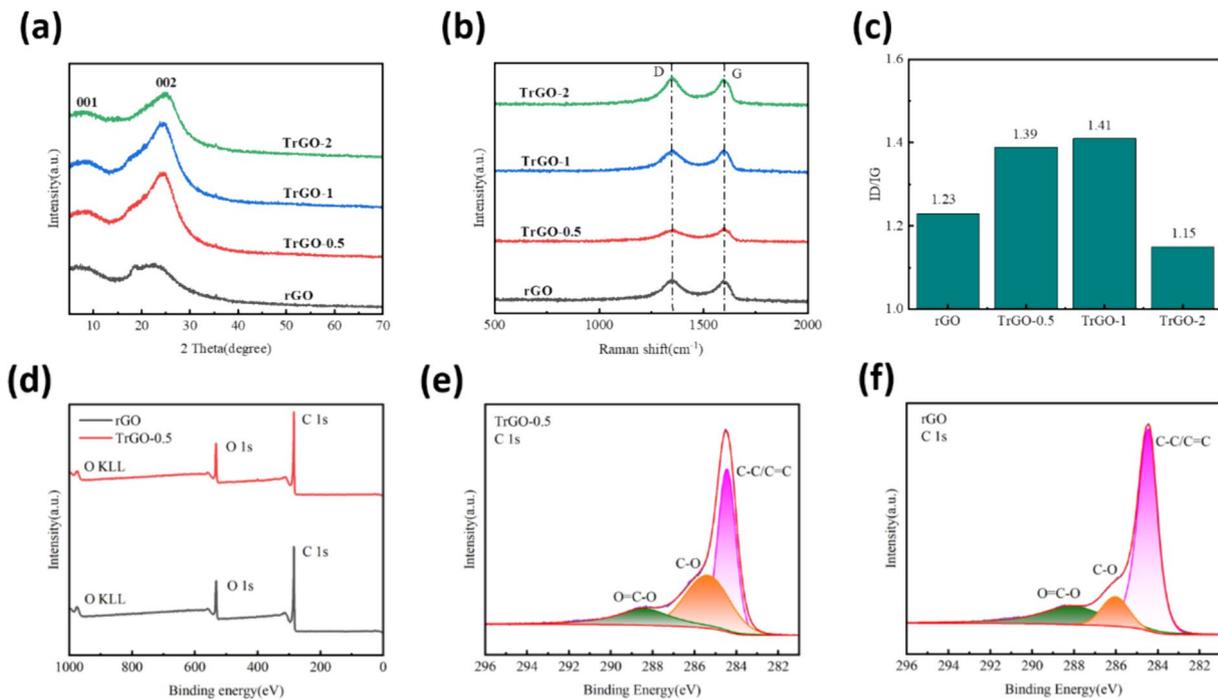


electrolyte. Figure 4a shows the CV curves of the rGO and TrGO electrodes at scan rate of  $200 \text{ mV s}^{-1}$ . Obviously, the introduction of TA does increase the specific capacitance of rGO, and TrGO-0.5 has the largest integrated area of the CV curve. The GCD results at current density of  $0.5 \text{ mA cm}^{-2}$  also show that TrGO-0.5 has the highest specific capacitance among the samples (Fig. 4b). TrGO-1 and TrGO-2 have lower specific capacitance than that of TrGO-0.5, possibly because the higher content of TA in TrGO-1 and TrGO-2 would lead to partial blockage of some micropores in rGO, resulting in lower transfer ability of electrolyte ions in TrGO-1 and TrGO-2. However, if the content of TA is too low, the effect of TA would not fully play. Therefore, TrGO-0.25 also has lower specific capacitance than that of TrGO-0.5, which can be observed from the CV and GCD curves (Fig. S1). The CV and GCD curves of TrGO electrodes illustrate their main EDL capacitance accompanied by partial pseudocapacitance.

The CV curves of TrGO-0.5 at scan rates from 10 to  $500 \text{ mV s}^{-1}$  are shown in Fig. 4c, and its GCD curves at current densities from  $0.5$  to  $8 \text{ mA cm}^{-2}$  are displayed in Fig. 4d.

Based on the GCD discharge curves, the areal capacitance and mass capacitance of rGO and TrGO-0.5 (Fig. 4e and f) are calculated according to the formula in supporting material (S1, S2). The areal capacitances of rGO and TrGO-0.5 are  $434$  and  $525 \text{ mF cm}^{-2}$  at  $0.5 \text{ mA cm}^{-2}$  (Fig. 4e), and their mass capacitances are  $134$  and  $147 \text{ F g}^{-1}$  at  $0.2 \text{ A g}^{-1}$  (Fig. 4f), respectively. Table 1 presents the performance comparison of TrGO-0.5 with some reported rGO-based materials [47–53], such as the rGO/ $\text{Ti}_3\text{C}_2\text{T}_x$  composite ( $115 \text{ F g}^{-1}$  at  $0.2 \text{ A g}^{-1}$ ) [48] and the rGO/ZnO composite ( $60 \text{ F g}^{-1}$  at  $5 \text{ mV s}^{-1}$ ) [53]. Appropriate TA content can not only promote the reduction of GO but also alleviate the stacking of rGO sheets and provide some redox active sites, resulting in the specific capacitance increase of TrGO-0.5.

The analysis of EIS data would reveal the information regarding ion diffusion resistance and transport ability. Figure 4g shows the Nyquist plots of rGO and TrGO-0.5 (The equivalent circuit diagram of TrGO-0.5 electrode is shown in Fig. S2). In the low frequency region, the higher curve slope of TrGO-0.5 compared to that of rGO indicates that TrGO-0.5 has better capacitive response than rGO. Therefore, the



**Fig. 3** **a** XRD spectra of rGO and TrGO samples. **b, c** Raman spectra and ID/IG of rGO and TrGO samples. **d** XPS spectra of rGO and TrGO-0.5. **e, f** High resolution of the C1s spectra of TrGO-0.5 and rGO

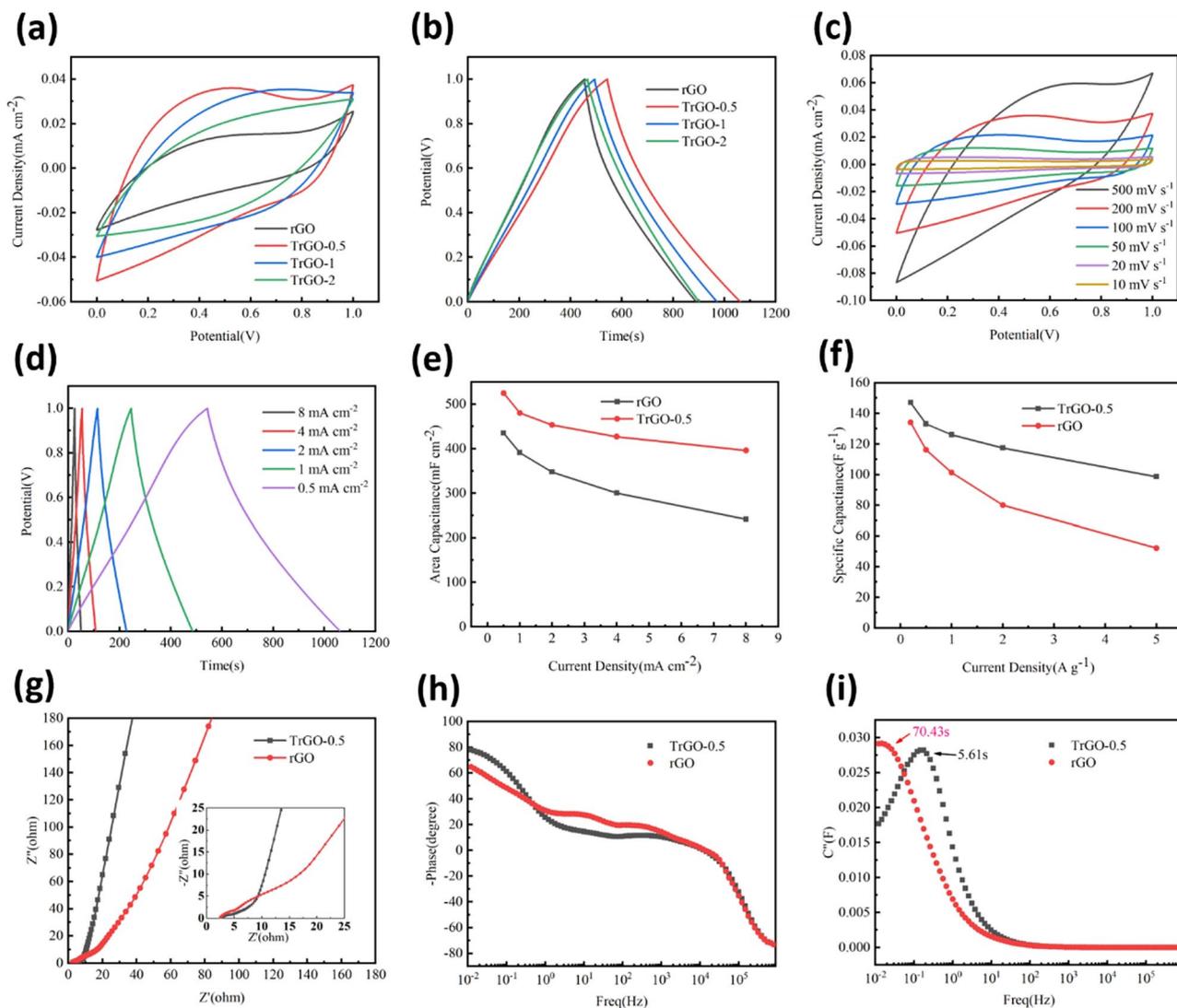
introduction of TA indeed enhances the ions diffusion and transfer kinetics in TrGO-0.5. At high-frequency region (the insert in Fig. 4g), the intersection of Nyquist plot with the x-axis corresponds to the equivalent series resistance (ESR), which includes the intrinsic resistance of the electrode and collector, the diffusion resistance of the electrolyte, and the contact resistance at electrode-electrolyte interface [54]. Both rGO and TrGO-0.5 exhibit low ESR which is about 2  $\Omega$ . The Bode phase plots (Fig. 4h) show that the phase angle of rGO at low frequencies is 65.9°, while that of TrGO-0.5 increases to 78.3°. Therefore, the phase angle of TrGO-0.5 is closer to 90° compared to that of rGO, which also indicates that TrGO-0.5 has more ideal capacitive behavior. According to the formula

of  $C'' = Z' / 2\pi f |Z|^2$  [55], in which  $C''$ ,  $Z'$  and  $Z$  are the virtual capacitance, the imaginary part and the modulus value of the impedance, respectively, the virtual capacitances of rGO and TrGO-0.5 are calculated and plotted versus frequency (Fig. 4i). The time electrostatic constant ( $\tau = 1/f_0$ ) of rGO and TrGO-0.5 are 70.43 and 5.61s, respectively, where  $f_0$  is the frequency corresponding to the maximum virtual capacitance. The much shorter  $\tau$  of TrGO-0.5 indicates that it has faster frequency response and ion transfer rate, which is consistent with the results shown in the Nyquist and Bode plots.

To further investigate the charge storage mechanism of the electrodes, the relationship between peak current ( $i$ ) and scan rate ( $v$ ) is calculated according to the

**Table 1** Performance comparison of this work with some reported results

Electrode material	Electrolyte	Capacitance	Electrochemical testing	ref
RGO/TA	1 M H <sub>2</sub> SO <sub>4</sub>	238.74mF cm <sup>-2</sup> at 1 mA cm <sup>-2</sup>	Three electrodes in Ag/AgCl	[42]
RGO/SiVW <sub>11</sub>	0.5 M H <sub>2</sub> SO <sub>4</sub>	377.4mF cm <sup>-2</sup> at 1.5 mA cm <sup>-2</sup>	Button cell	[49]
RGO/Ti <sub>3</sub> C <sub>2</sub> Tx	6 M KOH	115F g <sup>-1</sup> at 0.2 A g <sup>-1</sup>	Symmetrical two-electrode	[50]
RGO/PPy	1 M H <sub>2</sub> SO <sub>4</sub>	631mF cm <sup>-2</sup> at 1mA cm <sup>-2</sup>	An Ag/AgCl as reference electrode	[51]
RGO/CNT/ANF	6 M KOH	79F g <sup>-1</sup> at 200mv s <sup>-1</sup>	Button cell	[52]
RGO/AC/CNT	1 M LiClO <sub>4</sub>	101F g <sup>-1</sup> at 0.2A g <sup>-1</sup>	Button cell	[53]
RGO/MnCO <sub>3</sub>	1 M Na <sub>2</sub> SO <sub>4</sub>	468mF cm <sup>-2</sup> at 2.8mA cm <sup>-2</sup>	Three-electrode cells	[54]
RGO/ZnO	1 M KCl	60.63F g <sup>-1</sup> at 5mv s <sup>-1</sup>	Three-electrode cells	[55]
This work	6 M KOH	520mF cm <sup>-2</sup> at 0.5mA cm <sup>-2</sup> /147F g <sup>-1</sup> at 0.2A g <sup>-1</sup>	Symmetrical two-electrode	

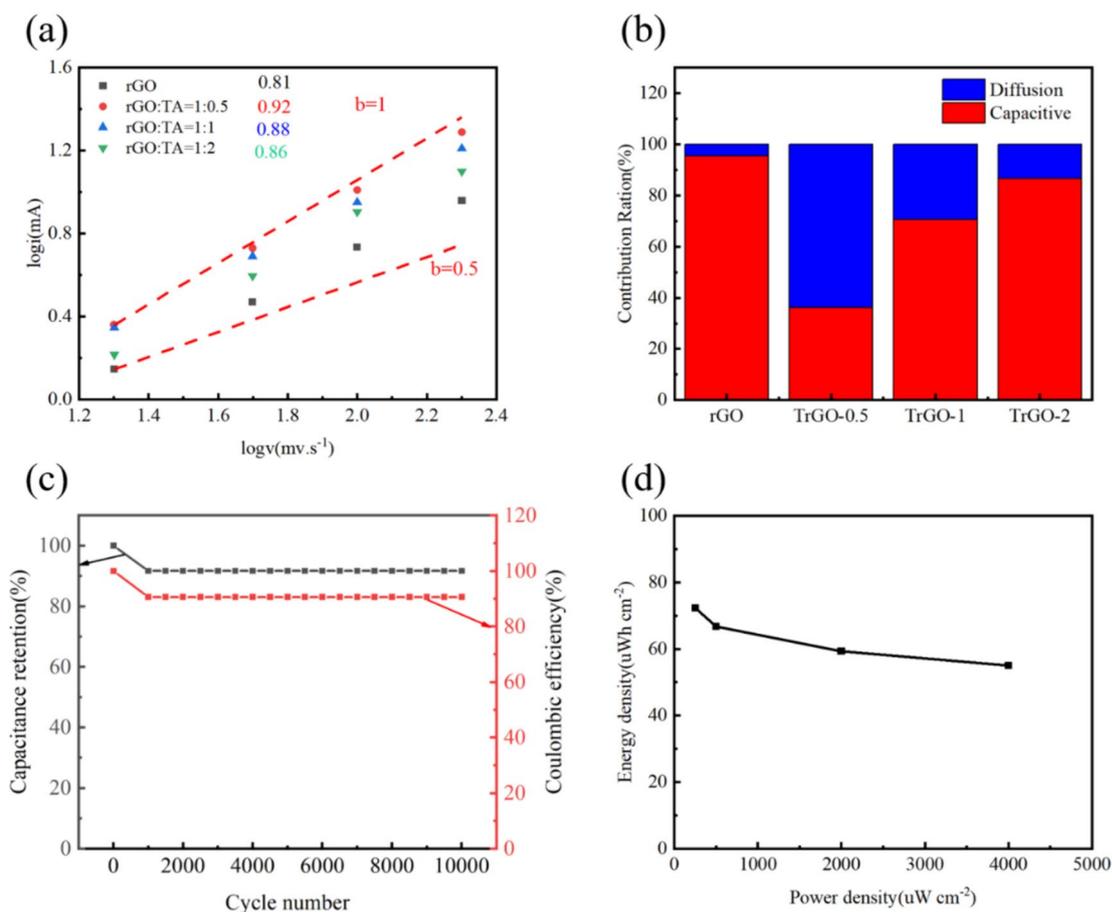


**Fig. 4** **a** CV curves of rGO and TrGO electrodes at scan rate of  $200 \text{ mV s}^{-1}$ . **b** GCD curves of rGO and TrGO electrodes at current density of  $0.5 \text{ mA cm}^{-2}$ . **c** CV curves of TrGO-0.5 at different scan rates. **d** GCD curves of TrGO-0.5 at different current densities. **e** Area capacitance of rGO and TrGO-0.5 at different current densities. **f** Mass capacitance of rGO and TrGO at different current densities. **g** Nyquist plots

of rGO and TrGO-0.5, the insert is the enlarged high-frequency region. **h** Bode plots of phase angle versus frequency of rGO and TrGO. **i** Relationship between imaginary capacitance and frequency of rGO and TrGO-0.5. The electrochemical performances were all tested in two-electrode symmetrical cells in  $6 \text{ M KOH}$  electrolyte

formulai =  $av^b$ , where parameters  $a$  and  $b$  are adjustable. The  $b$  value of  $0.5$  indicates surface-controlled behavior, while the  $b$  value of  $1$  represents diffusion-controlled process. The  $b$  values for rGO, TrGO-0.5, TrGO-1 and TrGO-2 are calculated to be  $0.81$ ,  $0.92$ ,  $0.88$  and  $0.86$ , respectively, as shown in Fig. 5a. The results suggest that all electrodes exhibit a combination of diffusion-controlled and surface-controlled capacitance behavior, and the introduction of TA leading to more influence of diffusion process. Furthermore, the contributions of surface

capacitance and diffusion capacitance can be calculated by equation (S9) (Fig. 5b). The specific capacitance of rGO is primarily composed of surface capacitance, with small portion of diffusion capacitance. However, with the introduction of TA, the diffusion capacitance increases. This indicates that TA can additionally enhance pseudocapacitive contribution, when reaching its maximum in TrGO-0.5 electrode. However, excessive introduction of TA may lead to the decrease of the active sites, and cause the surface micropores of rGO to be



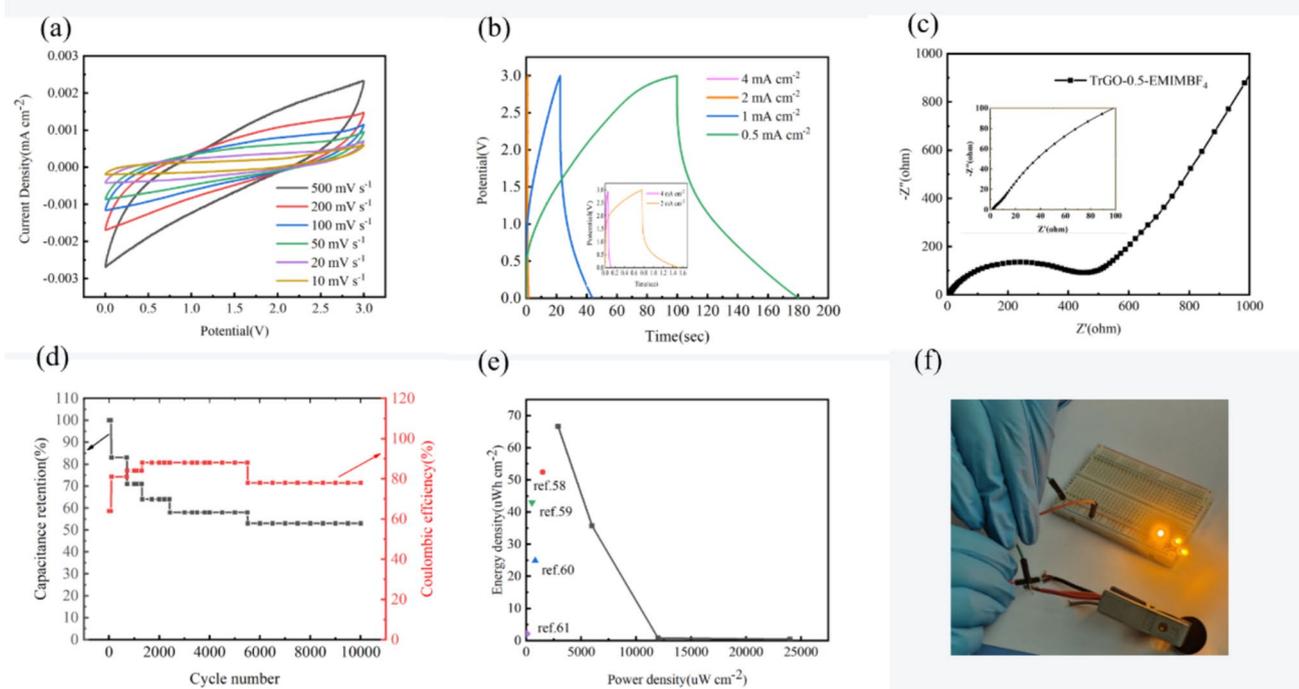
**Fig. 5** **a** Logarithmic plot of peak current as a function of scan rate. **b** Capacitance contribution of the rGO and TrGO electrodes at the scan speed of  $20 \text{ mV s}^{-1}$ , the red area represents the surface capacitance, and the blue area represents the diffusion capacitance. **c** Cycling per-

formance and Coulombic efficiency of TrGO-0.5 tested at  $5 \text{ A g}^{-1}$ . **d** The plot of energy density with power density (Ragone plot) of TrGO-0.5

blocked, resulting in the decrease of diffusion capacitance. Therefore, from TrGO-0.5 to TrGO-1 and TrGO-2, their diffusion capacitance keeps decreasing. TrGO-0.5 also shows excellent cycling stability and Coulombic efficiency. After 10,000 cycles at current density of  $4 \text{ mA cm}^{-2}$ , its capacitance retention and Coulombic efficiency are 91.7% and 90.6%, respectively (Fig. 5c). The energy density of TrGO-0.5 can reach  $72.2 \text{ uWh cm}^{-2}$  at power density of  $250.9 \text{ uW cm}^{-2}$ . When the power density was increased to  $4000.6 \text{ uW cm}^{-2}$ , TrGO-0.5 still has high energy density of  $55.03 \text{ uWh cm}^{-2}$  (Fig. 5d).

For the practical application, TrGO-0.5 was also assembled to button cell in  $2 \text{ M EMIMBF}_4$  electrolyte, its potential window can reach  $3 \text{ V}$  [14] (Fig. 6). Both CV (Fig. 6a) and GCD (Fig. 6b) curves indicate the good capacitive behavior of TrGO-0.5 in  $\text{EMIMBF}_4$  electrolyte. Nyquist plot shows the

about  $45^\circ$  slope at low-frequency region (Fig. 6c) and ESR of  $4.5 \Omega$  (insert in Fig. 6c). The diameter of the semicircle in high-frequency region is related to the ion transfer resistance at electrode and electrolyte interface. It is about  $384.2 \Omega$  for TrGO-0.5 in  $\text{EMIMBF}_4$  (Fig. 6c), significantly higher than that in  $6 \text{ M KOH}$  (the insert in Fig. 4g). Additionally, we also tested the cyclic stability and Coulombic efficiency of TrGO-0.5 based button cell (Fig. 6d). After 10,000 charge/discharge cycles at  $2 \text{ mA cm}^{-2}$ , the capacitance retention and Coulombic efficiency are 53% and 90.1%, respectively. Ragone plot shows that TrGO-0.5 has decent energy density of  $10 \text{ Wh kg}^{-1}$  at power density of  $300.2 \text{ W kg}^{-1}$ , which is superior than some recent reported carbon based materials [47, 56–58] (Fig. 6e). This TrGO-0.5 based button cell can light up three LEDs on a breadboard (Fig. 6f), which indicates its potential for practical application.



**Fig. 6** Electrochemical performance of TrGO-0.5 based button cell in 2 M EMIMBF<sub>4</sub> electrolyte. **a** CV curves at different scan rates. **b** GCD curves at different current densities. **c** Nyquist plot. **d** Cycle sta-

bility and Coulombic efficiency tested at 4 mV cm<sup>-2</sup>. **e** Ragone plot. **f** Illustration of TrGO-0.5 based button cell to light up three LEDs

## Conclusion

In summary, we synthesized TrGO layered films by modifying the rGO sheets with TA molecules for supercapacitor, resulting in enhanced areal electrochemical performance. Specifically, when tested in two-electrode symmetrical cell with 6 M KOH electrolyte, the TrGO-0.5 gives area capacitance of 525 mF cm<sup>-2</sup> at 0.5 mA cm<sup>-2</sup>, energy density of 72.2 uWh cm<sup>-2</sup> at power density of 250.9 uW cm<sup>-2</sup>. It also has excellent cycling stability of 91.7% retention after 10,000 charging/discharging cycles at current density of 4 mA cm<sup>-2</sup>. TrGO-0.5 was also assembled to button cell in 2 M EMIMBF<sub>4</sub> electrolyte for practical application, and it can light up three LEDs. Our study provides experimental evidence for the preparation of TrGO electrodes by introducing TA to modify the rGO sheets for practical application.

**Supplementary Information** The online version contains supplementary material available at <https://doi.org/10.1007/s10008-024-05946-y>.

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